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# Monolithic AlGaAs second-harmonic nanoantennas

V. F. GILI,<sup>1</sup> L. CARLETTI,<sup>2</sup> A. LOCATELLI,<sup>2</sup> D. ROCCO,<sup>2</sup> M. FINAZZI,<sup>3</sup> L. GHIRARDINI,<sup>3</sup> I. FAVERO,<sup>1</sup> C. GOMEZ,<sup>4</sup> A. LEMAÎTRE,<sup>4</sup> M. CELEBRANO,<sup>3</sup> C. DE ANGELIS,<sup>2</sup> AND G. LEO<sup>1,\*</sup>

<sup>1</sup>*Matériaux et Phénomènes Quantiques, Université Paris Diderot - Sorbonne Paris Cité, 10 rue A. Domon et L. Duquet, 75013 Paris, France*

<sup>2</sup>*Dept. of Information Engineering, University of Brescia, Via Branze 38, 25123 Brescia, Italy*

<sup>3</sup>*Dept. of Physics, Politecnico di Milano, Piazza Leonardo Da Vinci 32, 20133 Milano, Italy*

<sup>4</sup>*Laboratoire de Photonique et Nanostructures, CNRS-UPR20, Route de Nozay, 91460 Marcoussis, France*

\*giuseppe.leo@univ-paris-diderot.fr

**Abstract:** We demonstrate monolithic aluminum gallium arsenide (AlGaAs) optical nanoantennas. Using a selective oxidation technique, we fabricated epitaxial semiconductor nanocylinders on an aluminum oxide substrate. Second harmonic generation from AlGaAs nanocylinders of 400 nm height and varying radius pumped with femtosecond pulses delivered at 1554-nm wavelength has been measured, revealing a peak conversion efficiency exceeding  $10^{-5}$  for nanocylinders with an optimized geometry.

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**OCIS codes:** (130.3120) Integrated optics devices; (050.6624) Subwavelength structures; (190.2620) Harmonic generation and mixing.

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## 1. Introduction

Metal-less nanophotonics has recently gained an increasing interest because the optical response of high permittivity dielectric nanoparticles in a low refractive index background exhibits negligible dissipative losses and strong magnetic multipole resonances [1–3] both in the visible and near-infrared regions of the electromagnetic spectrum. These optical properties can be exploited in many applications such as directional scattering and emission [4–7], surface enhanced vibrational spectroscopy [8,9], and flat metasurfaces for phase-front engineering [10–12]. Compared to metallic nanoparticles, where the electric field is strongly confined close to the metal surface, the electric field of the resonant modes in dielectric nanoparticles penetrates deep inside their volume, strongly enhancing intracavity light-matter interactions. This fact encourages the use of all-dielectric nanoantennas for strengthening the nonlinear optical response, and some promising results have been achieved in the last two years in the framework of semiconductor technology. In particular, third-harmonic generation (THG) in silicon-on-insulator (SOI) nanoantennas has been investigated [13–15]. By using an array pattern of silicon nanostructures, a THG conversion efficiency higher than  $10^{-6}$  was demonstrated, which is the highest value reported for THG on nanoscale structures using comparable pump energies [15]. Another experimental demonstration showed an increase of two-photon absorption (TPA) by almost two orders of magnitude in hydrogenated amorphous silicon nanodisks with respect to an unstructured silicon film [16]. Although these results clearly illustrate the potential of all-dielectric nanoparticles for nonlinear nanophotonic applications, they are limited by the inherent constraints of the SOI platform in which they have been obtained.

Let us recall that the SOI technology, which is well known for large-scale production of electronic devices, is also becoming a standard for integrated photonics [17]. Its building block is a planar shallow waveguide that strongly confines light thanks to the high refractive

index contrast between a silicon (Si) core and a silicon-dioxide ( $\text{SiO}_2$ ) substrate ( $n_{\text{Si}} \approx 3.45$  and  $n_{\text{SiO}_2} \approx 1.44$  at a wavelength around  $1.5 \mu\text{m}$ ). Despite the spectacular achievements of silicon technology, the perspective of monolithic SOI photonics seems hardly viable because of a few hard facts: 1) silicon is not a laser material due to its indirect energy gap; 2) it is affected by TPA at optical telecom wavelengths due to its gap of  $1.1 \text{ eV}$ ; 3) it does not exhibit quadratic optical nonlinearity, because of its centro-symmetric crystal structure. Owing to the latter two reasons, only  $\chi^{(3)}$  nonlinear effects have been observed so far in dielectric nanoantennas. In order to overcome on the one hand these limitations, and on the other hand the technological burden of the wafer bonding step associated to the recently reported hybridization of AlGaAs on the SOI system [18], here we propose an AlGaAs-based monolithic platform for nonlinear nanophotonics.

III-V semiconductor  $\text{Al}_x\text{Ga}_{1-x}\text{As}$  is a popular material for integrated nonlinear optics [19] because, besides a few properties that it shares with Si (a similar refractive index in the near infrared, a  $\chi^{(3)}$  nonlinear index, a good thermal conductivity), it also exhibits some key comparative advantages: 1) a direct gap which increases with Al molar fraction  $x$  ( $E_g = 1.424 + 1.266x + 0.26x^2 \text{ eV}$ , for  $x < 0.45$ ) and enables TPA-free operation at  $1.55 \mu\text{m}$  for  $x \geq 0.18$ ; 2) a huge non-resonant quadratic susceptibility ( $d_{14} \approx 100 \text{ pm/V}$  for GaAs in the near infrared); 3) a wide transparency window in the mid-infrared (up to  $17 \mu\text{m}$ ); and 4) a mature heterostructure laser technology. However, a full development of an AlGaAs photonic platform has been hindered so far by the difficulty of fabricating monolithic shallow nanowires as in the SOI system, although the selective oxidation of AlGaAs epitaxial layers was discovered in 1990 and gives rise to an aluminum oxide ( $\text{AlOx}$ ) with optical and electrical properties similar to  $\text{SiO}_2$  [20]. Let us point out that while the use of  $\text{AlOx}$  layers thinner than  $100 \text{ nm}$  is standard in VCSEL technology and has resulted in the demonstration of an optical parametric oscillator in an AlGaAs waveguide [21], the fabrication of devices with high-quality micron-thick  $\text{AlOx}$  optical substrates has not been reported to date.

In this paper we first describe the fabrication of  $\chi^{(2)}$  AlGaAs-on- $\text{AlOx}$  monolithic nanostructures on a GaAs wafer, which we carried out by an appropriate design of the epitaxial structure and a careful control of the oxidation parameters (Section 2). Then, based on numerical modeling, we predict the occurrence of second harmonic generation (SHG) in  $\text{Al}_{0.18}\text{Ga}_{0.82}\text{As}$  nanocylinders on  $\text{AlOx}$ , with a conversion efficiency up to  $7 \times 10^{-5}$  for a  $1.6 \text{ GW/cm}^2$  pump in the optical telecom wavelength range (Section 3). Finally, in Section 4 we report and discuss the first experimental demonstration of the nonlinear response of AlGaAs-on- $\text{AlOx}$  nanoantennas, in fair agreement with such predictions.

## 2. Fabrication

The selective oxidation of AlGaAs optical substrates requires some care, in order to avoid that the oxidation-induced contraction results in defects at the interfaces between the non-stoichiometric  $\text{AlOx}$  and the adjacent crystal lattices. Our samples were grown by molecular-beam-epitaxy on [100] non-intentionally doped GaAs wafer, with a  $400 \text{ nm}$  layer of  $\text{Al}_{0.18}\text{Ga}_{0.82}\text{As}$  on top of an aluminum-rich substrate to be oxidized at a later stage [22]. In order to improve the eventual adhesion between  $\text{AlOx}$  and the adjacent crystalline layers, such substrate consists of a  $1\text{-}\mu\text{m}$ -thick  $\text{Al}_{0.98}\text{Ga}_{0.02}\text{As}$  layer sandwiched between two  $90\text{-nm}$ -thick layers with linear tapers of Al molar fraction. To obtain an array of nanodisks as shown in Fig. 1(a), we patterned circles with radii between  $180 \text{ nm}$  and  $220 \text{ nm}$ , and equally spaced by  $3 \mu\text{m}$ , with a scanning electron microscope (SEM) lithography system. Then our samples were dry etched with non-selective ICP-RIE plus  $\text{SiCl}_4\text{:Ar}$  chemical treatment. The etching depth of  $400 \text{ nm}$ , controlled by laser interferometer, defined the nanocylinders and exposed the  $\text{Al}_{0.98}\text{Ga}_{0.02}\text{As}$ . Then, the etched sample was oxidized at  $390^\circ\text{C}$  for 30 minutes in an oven equipped with in situ optical monitoring, under a precisely controlled water vapor flow with  $\text{N}_2\text{:H}_2$  gas carrier. After oxidation, each  $\text{Al}_{0.18}\text{Ga}_{0.82}\text{As}$  nanocylinder lies upon a uniform  $\text{AlOx}$

substrate (see Fig. 1(b)), whose low refractive index ( $n \approx 1.6$ ) enables sub-wavelength optical confinement in the nanocavity by total internal reflection.

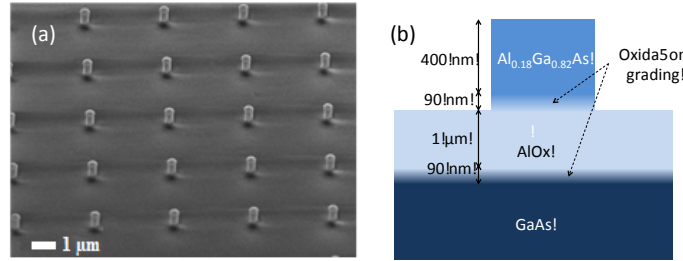


Fig. 1. Monolithic AlGaAs-on-AlOx nanoantennas: (a) scanning-electron-microscope picture of a part of the array; (b) schematics of a single nanoantenna.

### 3. Numerical modeling

The linear and nonlinear optical response of the disk nanoantennas on an aluminum-oxide (AlOx) substrate are modeled by using frequency-domain finite element simulations. The scattering efficiency spectra as a function of disk radius are shown in Fig. 2(a): we can observe that the resonance in this wavelength range red-shifts while the radius is increased [23]. In Fig. 2(b) the scattering efficiency spectra of a disk with a radius of 200 nm is obtained by extracting the data along the horizontal dashed line on Fig. 2(a).

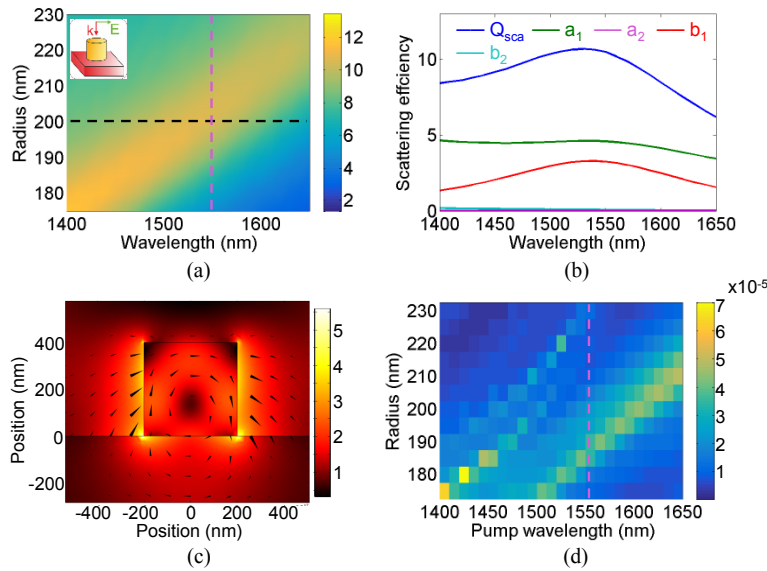


Fig. 2.: (a) Scattering efficiency spectra as a function of the nanoantenna radius. The horizontal dashed line represents an antenna with radius  $r = 200$  nm while the vertical dashed line represents the wavelength of 1550 nm. The schematics of the illumination is shown in the inset. Light is at normal incidence on the nanoantenna with electric field  $E$  linearly polarized parallel to the substrate plane. (b) Scattering efficiency ( $Q_{sca}$ ) spectra for a disk with  $r = 200$  nm, with the multipole expansion terms of the optical response. The coefficients  $a_1$ ,  $b_1$ ,  $a_2$ , and  $b_2$  are due to electric dipole, magnetic dipole, electric quadrupole, and magnetic quadrupole contributions, respectively. (c) Electric field enhancement map in the incidence plane of a disk with  $r = 200$  nm at the MD resonance ( $\lambda = 1540$  nm). Black cones represent the electric field vector. (d) SH conversion efficiency as a function of the pump wavelength and nanoantenna radius. The vertical dashed line represents the wavelength of 1550 nm. All results are obtained by using a fixed nanoantenna height of 400 nm.



In order to gain a deeper physical insight on the behavior of the nanoantenna, the optical response is expanded using a multipole decomposition. Because of the presence of the substrate, exact values of the coefficients associated to the various multipole contributions are obtained via integration of the dielectric currents induced in the AlGaAs disk by the incident field [24,25]. Despite the numerical simulations take into account the presence of the substrate throughout all the paper, for simplicity, in Fig. 2(b) the cross-sections associated with each multipole are calculated as if it was radiating into air, thus the total cross-section is slightly higher than the sum over multipole contributions. As it can be seen from Fig. 2(b), the resonance around 1.55  $\mu\text{m}$  is due to a magnetic dipole (MD) mode and an electric dipole (ED) contribution. The MD character of the resonance is clearly seen from the E field map in Fig. 2(c), which also shows that the electric field loop that characterizes the MD mode partially leaks into the AlOx substrate [26].

The second-order nonlinear optical response associated with the SHG is modeled using the nonlinear polarization induced by  $\chi^{(2)}$  as a source in COMSOL simulations. Since AlGaAs has a zinc blende crystalline structure, the only non-vanishing terms of the nonlinear susceptibility tensor are the  $d_{ijk}$  terms with  $i \neq j \neq k$ , which, in the calculations, have a value of 100 pm/V [27]. The conversion efficiency is defined as

$$\eta_{SHG} = \frac{\int_A \vec{S}_{SH} \cdot \hat{n} da}{I_0 \times \pi r^2}$$

where  $\vec{S}_{SH}$  is the Poynting vector of the SH field,  $\hat{n}$  is the unit vector normal to a surface A enclosing the antenna, and  $I_0$  is the incident field intensity ( $I_0 = 1 \text{ GW/cm}^2$  in the simulations). Figure 2(d) shows the calculated SH conversion efficiency as a function of the pump wavelength and for varying disk radius. We numerically find a SH conversion efficiency up to  $6 \times 10^{-5}$  for a pump wavelength between 1400 nm and 1650 nm while the disk radius is varying between 175 nm and 230 nm. The SH conversion efficiency is smaller than the value reported in [23] due to the field leakage effect caused by the substrate that hampers the strength of the MD resonance. The SHG peaks in Fig. 2(d) arise when the resonant mode at the pump wavelength spatially overlaps with a mode at the wavelength of the corresponding SH emission. MD and ED resonances play a dominant role at the pump frequency; conversely at the SH frequency sharper resonances with high-order multipolar contributions are the main selection mechanism. By scanning the pump wavelength the spatial overlap between the modes varies and the SH conversion efficiency is consequently determined.

#### 4. Experimental results and discussion

We have experimentally investigated the nonlinear properties of individual fabricated nanopillars to validate the theoretical description presented above. We employed the nonlinear microscopy setup as sketched in Fig. 3, which allows achieving a lateral resolution of about 700 nm in the nonlinear regime (well-below the interparticle distance). The excitation light is delivered by a linearly polarized (extinction ratio  $\approx 100:1$ ) ultrafast Erbium-doped fiber laser centered at 1554 nm laser (150 fs pulse length). The light polarization is finely-tuned by means of a  $\lambda/2$  plate before entering the objective. The excitation beam is tightly focused onto the sample through a high-NA objective (Nikon, NA = 0.85) to allow the interrogation of individual pillars inside the array under study. The light emitted by the sample, collected through the same optical path, is reflected by a polarization-maintaining dichroic mirror (DMSP1000, Thorlabs Inc.) to reject the excitation light. To further clean up the nonlinear signal from possible two-photon photoluminescence (TPL) we inserted a short-pass filter (FESH 1000, Thorlabs Inc.) inside the detection line, which is sensitive from 500 to 1000 nm. The spectral components of the nonlinear emitted light are investigated using a visible/near-infrared grating-based spectrometer (Shemrock 303i, Andor Technology). An

alternative SHG signals were collected using the detection path including the SPAD from an array of nanocylinders with radius varying from 175 to 225 nm. The dependence of the experimentally detected SHG on the radius size of the nanocylinders is reported in Fig. 4(a) (hollow diamonds). The plotted values are subtracted by the background coming from the AlOx substrate, which is however about 3 orders of magnitude below the nanoantenna signal. A multi-Lorentzian fit of the curve allows identifying three features in the emission behavior.

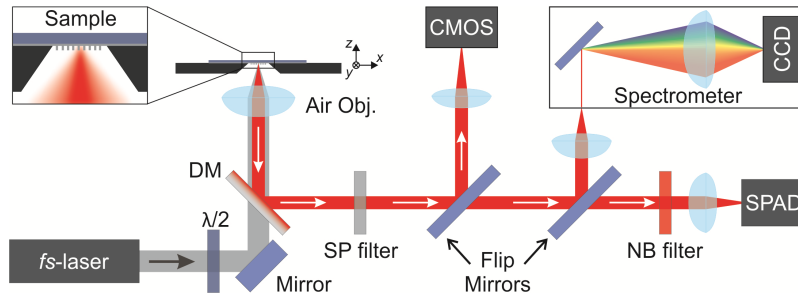


Fig. 3.: Experimental set-up for the nonlinear spectroscopy of individual nanoantennas. Fs-laser: 1554 nm Er-doped fiber laser delivering 150-fs pulses (light grey beam).  $\lambda/2$ : half-wave plate; Air Obj.: 0.85-NA air objective; DM: dichroic mirror; SP & NB filters: short-pass and narrow-band filter. CMOS: imaging camera; SPAD: single-photon avalanche detector; CCD: spectrometer cooled camera. The sample is mounted facing downwards as depicted in the inset.

The calculated radius-dependent behavior of the conversion efficiency reported in Fig. 4(b) singles out three resonances for the SHG nanocavity modes excited at three specific radii size, in striking agreement with the experimental values of the overall SHG collected signal. The small shift of about 5 nm that can be observed on the main peak of the curve is attributed to fabrication tolerances between the nominal and the actual size of the nanocylinders.

We have measured the SHG intensity emitted by the most efficient nanoantenna (193-nm radius) for an illumination power ranging from 5 to 750  $\mu\text{W}$ . Figure 4(c) reports the power-dependent curves of the SHG signal, retrieved from the signal recorded at the SPAD side. The full circles represent the SHG photon counts recorded while increasing the incident power level, whereas the hollow squares indicate the photon counts while decreasing the input power. The qualitatively good overlap between the two curves shows that no degradation occurs in this power interval, while the quadratic fit to the curves (dashed line) confirms the second order origin of this process.

Figure 4(c) reveals a peak count rate of about  $1.4 \times 10^8$  photons per second, which corresponds to an effective SHG photon flux of about  $3.2 \times 10^{10}$  per second leading to a peak conversion efficiency of about  $1.1 \times 10^{-5}$ . These figures are obtained by accounting for the SPAD filling-factor (0.34), the throughput of all the optical elements in the detection path including the objective (0.13), and the objective collection efficiency (0.1). The latter is evaluated by considering the SHG power fraction coupled to the objective acceptance cone assuming an isotropic radiation pattern. The uncertainty associated with the last coefficient limits the accuracy of the experimentally determined SHG efficiency, which is nevertheless of the same order of magnitude as the peak efficiency evaluated with our calculations.

By flipping the mirror in front of the SPAD it is possible to couple the collected light to the spectrometer and measure the emission spectrum in the visible/near-infrared range. Figure 4(d) shows the emission spectrum recorded after removing the short-pass filter and at an incident power of 500  $\mu\text{W}$ . Even at this relatively high pump power, multi-photon photoluminescence (MPL) is barely detectable and only a very weak THG peak is observed.

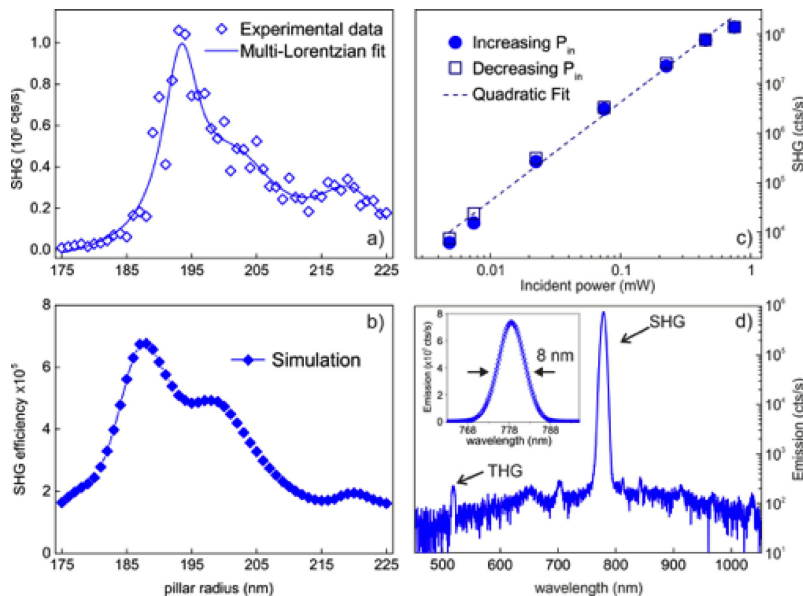


Fig. 4.: a) Measured SHG intensity rate as a function of the nanoantenna radius (hollow diamonds). As a guide to the eye a multi-Gaussian fit featuring three distinct peaks. b) Numerical calculations of the SHG efficiency on nanoantennas featuring the same geometrical parameters as the fabricated/measured ones. c) Power curve in Log/Log scale. SHG intensity as a function of the pump intensity for nanoantenna with 193 nm radius. d) Emission spectrum of the most efficient pillar acquired at 500  $\mu$ W incident power and plotted in Log scale. An intense SHG peak around 778 nm and a weak THG peak around 518 nm can be identified. Inset: a zoom of the main SHG peak plotted in linear scale (void diamonds) and a Gaussian fit to it (straight line), showing an 8-nm-width spectral feature.

It is fundamental here to stress that both MPL and THG signals are almost 4 orders of magnitude weaker than the main peak, which we attribute to pure SHG emission. This last claim is inherently supported by the peak spectral width, which is about  $\sqrt{2}$  times narrower than the excitation laser line (about 15 nm) as a consequence of a coherent two photon excitation process.

## 5. Conclusions

We have demonstrated monolithic AlGaAs optical nanoantennas fabricated on a high-quality, micrometer-thick AlOx optical substrate. These nanoantennas have been probed for SHG driven by a magnetic dipole resonance at the wavelength of 1550 nm. The experimental SHG spectra and conversion efficiency (about  $10^{-5}$  at  $1.6 \text{ GW/cm}^2$  pump intensity) are in excellent agreement with the numerical simulations. These results show the huge potential of AlGaAs nanoantennas for nonlinear nanophotonic applications.

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