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1 Crystallization of polarons through charge and spin ordering transitions in 1T-TaS₂

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10 ABSTRACT

11 The interaction of electrons with the lattice in metals can lead to reduction of their kinetic energy to the point 12 where they may form heavy, dressed quasiparticles – polarons. Unfortunately, polaronic lattice distortions are 13 difficult to distinguish from more conventional charge- and spin- ordering phenomena at low temperatures. 14 Here we present a study of local symmetry breaking of the lattice structure on the picosecond timescale in the 15 prototype layered dichalcogenide Mott insulator 1T-TaS₂ using X-ray pair-distribution function measurements. 16 We clearly identify symmetry-breaking polaronic lattice distortions at temperatures well above the ordered 17 phases, and record the evolution of broken symmetry states from 915 K to 15 K. The data imply that charge 18 ordering is driven by polaron crystallization into a Wigner crystal-like state, rather than Fermi surface nesting or 19 conventional electron-phonon coupling. At intermediate temperatures the local lattice distortions are found to 20 be consistent with a quantum spin liquid state.

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23 INTRODUCTION

Quasi-two dimensional quantum-ordered materials display very diverse and often exotic spin¹⁻³ and charge 24 ordering behaviour⁴, including exciton condensation⁵, chiral spin liquids⁶ and superconductivity^{7,8}. They can also 25 display hidden phases which can be manipulated by pressure 9,10 , light $^{11-15}$ or doping $^{16-18}$. An important aspect 26 27 that determines spin ordering behaviour is the competition between Coulomb repulsion and interlayer 28 interactions¹⁹. A common feature of these systems is that the strength of the electron phonon interaction (EPI) and electron density control the electron kinetic energy. Increasing the EPI leads to bandwidth narrowing, and 29 30 lattice dressed electron quasiparticles - as originally introduced by Landau in 1933²⁰. A further consequence of electron localization is an enhanced Coulomb interaction that leads to correlations, ubiquitous mesoscopic 31 32 textures^{21,22}, charge-density wave (CDW) formation²³ and superconductivity at high critical temperatures in the fluctuation-dominated intermediate coupling regime of layered cuprate superconductors²⁴. Recent model 33 calculations ²³ applied to wide range of transition metal dichalcogenide (TMD) materials and comparisons with 34 35 scanning tunneling microsopy experiments highlighted the importance of the Coulomb interaction that lead to ubiquitous charge ordering at magic filling fractions ²³. Specific features of the band structure may lead to further 36 37 effects, such as exciton condensation⁵ and CDW ordering aided by Fermi surface nesting²⁵ on top of the seemingly universal charge ordering²³. Experimental evidence for polarons comes from many different 38 techniques in a very wide variety of materials^{26,27}, but their role in forming quantum states, such as Wigner 39 40 crystals, superconductivity and quantum spin liquids is not experimentally clear, particularly on the transition 41 from the low to high density limit. Direct observation of symmetry-specific polaronic lattice distortions on the 42 low- to high- density limit are crucial for progress in understanding a wide class of quantum materials.

As a prototypical example, the layered dichalcogenide 1T-TaS₂ in equilibrium is of interest as a Mott insulator²⁸, 43 44 CDW system⁴, polaronic superconductor⁹ and guantum spin liquid². Additional recent interest in the material comes from the discovery of hidden light- or charge injection- induced mesoscopic metastable phases ^{12,29} that 45 are important both fundamentally and for applications30-32. Structurally, it is a kinetically trapped 1T polymorph 46 of TaS₂, with octahedral stacking (Fig 1a) ^{4,33–35}. It's deceptively simple electronic band structure in the 47 undistorted phase has a single Ta metal band crossing the Fermi level.^{12,34} While nominally metallic at high 48 temperatures, it shows an incommensurate (IC) CDW (ICCDW) along the principal crystal axes already at high 49 50 temperatures (between 550 and 350 K), with an associated Kohn anomaly at a wavevector $q_{IC} \simeq (0.283, 0.0)$ in units of the reciprocal lattice vector $a_0^{*36,37}$ which implies a strong coupling between electrons and the lattice. 51 52 Below 350 K, the IC state breaks up into a patchy periodic network of commensurate islands separated by discommensurations^{38–40}. Upon further cooling, this nearly commensurate (NC) CDW (NCCDW) state becomes 53 54 fully commensurate (C) CDW (CCDW) after a first order transition at ~ 180 K, with the appearance of a Mott gap in the charge excitation spectrum observed by single-particle tunneling⁴¹ and angle-resolved 55 56 photoemission^{25,42}. In contrast, gapless spin relaxation in the range 50-200 K is attributed to a possible quantum 57 spin liquid (QSL) phase, arising from the spin frustration on the triangular superlattice. While band structure 58 calculations³ suggest that the QSL phase may be suppressed by inter-layer coupling⁴³, this cannot explain the 59 marked difference of spin and charge excitation spectra². A further intriguing feature is a discontinuity in the 60 magnetic susceptibility and an anomalous peak in the spin-spin relaxation time concurrent with a crossover 61 around ~ 50 K of the spin-lattice relaxation time from a QSL-like, to a gapped T-dependence^{44,45}. Near 50 K, a 62 resistivity upturn from nearly T-independent resistivity to strongly T-dependent variable-range hopping behavior 63 has been commonly reported, suggesting an onset of low-temperature charge localization^{4,46}. The common 64 picture of the 1T-TaS₂ lattice structure in the C state assumes simple distortions from the average lattice 65 structure (trigonal P3m, Fig. 1a), whereby 12 Ta atoms are attracted symmetrically towards the central Ta atom 66 that carries an extra electronic charge, forming a star of David (SoD) pattern (Fig.1c). The SoDs are arranged in a $\sqrt{13} \times \sqrt{13}$ unit cell superlattice structure (P $\overline{3}$ symmetry⁴⁷). At such an electron density of 1/13, the 67 dimensionless ratio r_s of the Coulomb energy V to the kinetic (Fermi) energy E_K $r_s = \frac{V}{E_K} = \frac{e^2 m}{\hbar^2 \sqrt{n}} \simeq 70 \sim 100$, 68 69 where e is the elementary charge, n is the (2D) density, and $m \sim 3$ is the effective electron mass²⁵. Considering that the Wigner crystal stability limit is $r_s = 31 \sim 38^{23,48-50}$, 1T-TaS₂ is comfortably in the Wigner crystal regime, which means that electronic superlattice ordering on the basis of dominant Coulomb interactions and tell-tale lattice deformations around localized carriers may be anticipated. Room temperature extended X-ray absorption fine structure (EXAFS)⁵¹, high-resolution transmission electron microscopy (HRTEM)⁵²⁻⁵⁴ and X-ray structural measurements⁴⁴ indeed suggest the existence of symmetry-breaking atomic displacements, but studies over a wide range of ordering temperatures have not yet been performed.

Here we present the first systematic measurements of the local lattice structure that covers temperatures from 15 to 915 K. The data reveal local symmetry breaking from the established $P\bar{3}m$ symmetry at all measured temperatures (Fig. 1a, inset), and unambiguous evidence for individual polarons in the dilute high-temperature limit, revising our current notions about the origin of charge and spin ordering, with clear implications for other materials of current interest.

81 **RESULTS**

The local structure of 1T-TaS₂ is investigated using X-ray pair distribution function (PDF) analysis^{55,56}, giving picosecond-timescale structural snapshots for 15 K \leq T \leq 915 K. This encompasses the known electronically ordered states below 600 K portrayed by the resistivity vs. temperature plot shown in Fig. 1a⁹, as well as irreversible polytype transformations above 600 K.

86 The anisotropic atomic displacement parameters, U_{ii}, of Ta atoms were retrieved from a fit to the established 87 P3m structure over a range of 20-60 Å (Fig. 1b). For T < 600 K, the Uij, reveal an anomalously large in-plane 88 component, U_{11} , approximately a factor of 2 larger than the component along the stacking axis, U_{33} (e.g. at 500 K U₁₁ is 0.025 Å² whereas U₃₃ is 0.01 Å²). This is unusual for a layered system in which a larger out-of-plane 89 90 component is expected to arise from interlayer disorder. At high T, in the metallic (M) phase, the large U_{11} 91 signifies an apparent intralayer 'disorder', unaccounted for by the $P\overline{3}m$ structural model. At lower temperature, 92 the P $\overline{3}$ m model is inadequate at any temperature, revealed by the difference PDFs shown in Fig. 1c. The spatial 93 extent and character of this symmetry breaking changes across the cascade of CDW transitions, tracking complex 94 correlations in different electronic regimes. Intriguingly, as we describe below, nanoscale symmetry breaking is 95 also evident in the polymorphs above 630 K (Fig. 2b,c).

As an aid to understanding, pair-specific contributions to the total PDF within the $P\overline{3}m$ model are shown in Fig. 96 1e, where Ta-Ta pair contributions are dominant (see Methods). In the P $\overline{3}$ m structure, the undistorted Ta 97 98 sublattice features a single-valued nearest-neighbor Ta-Ta distance, seen as a sharp peak at ~3.4 Å, Fig. 1e. 99 However, the data reveal a distinct additional shoulder, (e.g. Figs. 1e and 2b for 600 K), which is unexplained by 100 the P3m model (Figs. 1c and 2c (top)). At high temperature this distortion is most likely dynamic, as expected 101 for symmetry-specific lattice fluctuations associated with carrier localization and polaron formation. Very 102 specific distributions of Ta-Ta and Ta-S pairs in the 3.3-5.9 Å range are expected for SoD polaronic distortions⁴⁷. In the C phase with a single layer $\sqrt{13} \times \sqrt{13}$ supercell P3 model (Methods and Supplementary Table 1) the SoD 103 polaron structure is adequately described (Fig. 2f)^{47,57}. However, for all other electronic regimes it was necessary 104 105 to add $P\overline{3}$ m as a secondary phase to achieve good fits (see Fig. 3c, inset). The temperature dependence of the 106 interatomic distances and bond angles, and deductions resulting from fits to the data are addressed sequentially 107 below, starting from high temperatures, well above any known charge or spin ordered phase.

Polymorphic regime – Heating 1T-TaS₂ above 600 K results in irreversible polymorphic transformations altering average symmetry and stacking, ^{26,51}where nominally 50% of octahedrally coordinated 1T layers convert into trigonal prismatically coordinated 1H layers ^{35,58} (Fig. 2a inset). This is accompanied by polymorph-specific stacking of the two layer-types ^{4,45} and changes of the local Ta environment in the prismatic layers. Two such transformations are seen in the PDF data (Fig. 2b), at 630 and 880 K (see Supplementary Figure 1 for more details). Remarkably, a shoulder-signal (~3.5 Å, Fig. 2b) associated with local lattice distortions (polarons) is

observable for 730 K < T < 915 K. The distorted fraction does not vanish in the polymorphs but instead drops to
half its value observed in the 1T regime (T < 630 K), with no significant reduction through the second
transformation (Fig. 2b). Notably, PDF data above 730 K conform to the 6R-TaS₂-like model (R3m symmetry)
featuring equally abundant alternating 1T and 1H layers (Fig. 2c, bottom) ⁵⁹. The model features undistorted 1T
and 1H layers, but a clear distortion signal, albeit weaker, such as is seen at 600 K, persists in the fit differential,
implying the existence of polaron-specific deformations in the 1T layers well above 600K (Fig. 2c, top), with ~10
wt% distorted fraction (see below).

121 High-temperature metallic regime – the data at 600 K show multiple components at the position where the 122 P3m model predicts a single Ta-Ta peak (Fig. 1c and red trace fit in Fig. 2c, R_w=15%). The distortion spans ~5 Å 123 with weaker signal extending up to ~12 Å, depicted by the difference PDFs. Such signatures, also seen in the IC 124 phase, are weaker than in the NC and C regimes (Fig. 1c). However, their similarity and continuity imply a 125 common origin and unambiguously reveals the presence of high temperature electron localization, detectable by charge-lattice coupling^{60,61}. Despite similar P3m model misfits shown in Fig. 1c, explicit data differences, Fig. 126 127 1d, reveal subtle changes across the M-IC transition for r > 5 Å. The electron localization results in incomplete 128 breakup of the P3m matrix, but without the formation of the full SoD motif. The light red profile in Fig. 2d and 129 the intensity band in Fig. 1e at ~3.3 Å remain broad, consistent with incoherent distortions from weakly 130 developed fluctuating charge correlations. In this diluted limit local distortions were approximated as P $\overline{3}$ 131 nanoscale inclusions, Fig. 2g, within the undistorted P3m matrix. This model (cyan traces in Fig. 2c (top), Rw=8%) accounts for the misfit, resulting in \sim 20 wt % of P $\overline{3}$ distortions involving heavily puckered hexagonal Ta discs 132 surrounding central tantalum. (This is consistent with ~10 wt% of locally distorted environments in the 1T-layers 133 134 in the polymorphic regime above 630 K discussed in the previous section).

135 **Incommensurate regime** – The growth of charge correlations and structural coherence of associated distortions 136 results in their signature extending over a longer length-scale (Fig. 1c). Stacks of experimental PDFs for 15-500 K (Fig. 2d,e) show that short length-scale peaks (r < 5 Å) sharpen gradually on cooling. However, the data still 137 138 does not show a clear SoD motif (Fig. 2d). The data at ~3.5 Å are broad and unresolved, and at ~3.75 Å are rather 139 featureless. Conversely, peaks at higher distances show the opposite temperature trend. In the light red stack, 140 Fig. 2e, the signal at ~23.5 Å is the sharpest just below the M-IC transition, portraying high symmetry, and systematically broadens with intensity reduction as temperature decreases, evidencing growth of broken-141 142 symmetry correlations. The change in slope of Te U₁₁(T) at 550 K (Fig. 1b) indicates an increased localized charge 143 density. Model-derived Ta-Ta distances for T > 350 K shown in Fig. 3a portray substantially distorted hexagonal 144 SoD cluster cores comprised of short Ta-Ta contacts (Fig. 3d,f). The distorted fraction increases to ~30 wt% compared to the M state (Fig. 3c). The PDF peaks which are sensitive to interlayer stacking are very broad, as 145 146 shown in Fig. 4 a,c, implying inhomogeneous stacking. The distribution centroids of the closest interlayer 147 separation, Fig. 4a, are shifted to the left, well below 5.9 Å, indicating local compression of layers which are, on 148 average, closer together than they would be in the undistorted 1T structure. The observation of negative thermal expansion jumps along the stacking axis is relevant in this context⁶² (see Supplementary Figures 2 and 3 for 149 150 discussion).

151 Nearly commensurate regime – Significant local structure changes occur at the IC-NC transition. The distortion 152 fingerprint, Fig. 1c, grows dramatically and PDF intensities vividly redistribute below 5 Å (Figs. 1f and 2d). The 153 Ta-Ta peak at 3.25 Å, which is broad in the IC phase, splits into an incompletely resolved triplet of unequal 154 intensity peaks at ~3.15 Å, ~3.3 Å and ~3.8 Å, marking the formation of well-defined SoD clusters. The feature 155 at ~3.15 Å corresponds to the Ta13 SoD intra-cluster configuration, whereas the other two features depict the 156 inter-cluster correlations (Fig. 3). Residual intensity at ~3.4 Å is associated with discommensuration regions of approximately $P\overline{3}m$ character, separating hexagonal domains of well-ordered SoD clusters within the Ta 157 layers^{63,64}, well observable by STM. Multiphase modeling with coexisting P3 and P3m phases confirms this 158 159 picture (Fig. 3c inset, see Methods). As SoD clusters emerge, the distortion magnitude increases (Fig. 3a,b), 160 consistent with increasingly resolved signal seen for 3 Å < r < 4 Å shown in Fig 2d. Importantly, the star vertices 161 exhibit sudden asymmetric contraction, accompanied by substantial puckering (Fig. 3d,f) and SoD twisting (Fig. 162 3e). The distorted P3 phase becomes the dominant fraction below 350 K, at ~60 wt% (Fig. 3c). This is consistent 163 with the patchy nature of the discommensurate NCCDW phase. Narrow thermal hysteresis, shown in Figs. 3 b-164 e, reflects the 1st order character of the IC-NC transition. As SoD local order is established in the NC state, they 165 simultaneously form well defined bilayer correlations, Figs. 4 a,c: The SoDs in adjacent layers couple and align 166 along the c-axis. In contrast, inter-bilayer correlations are still fuzzy: different bilayers exhibit appreciable 167 stacking disorder, evident from inter-bilayer sensitive peak, Figs. 4 b,d, which probes off-axis correlations whose 168 distribution is still considerably broad in the NC state.

169 Commensurate regime – At the NC-C transition, the PDF intensity at ~3.4 Å diminishes, implying the gradual 170 disappearance of the ordered domain walls, while the intra-SoD-cluster peak sharpens (Fig. 2d), consistent with 171 increased SoD density and additional charge localization. Similar changes are observed in the far-r limit (Fig. 2e), 172 as a homogeneous crystal structure forms in the commensurate state. Closer inspection of the data unveils an 173 additional restructuring below ~50 K, embodied in subtle peak shifts and sharpening observed in the dark blue 174 traces in Figs. 2d and 4a. Modeling shows that the distribution of Ta-Ta distances depicting intra- and inter-SoD 175 Ta-Ta contacts changes character in the C phase, with tightly bound SoDs structurally further regularizing below 50 K (Fig. 3a,b). The intra-SoD angle θ reduces, approaching 120° at base temperature, Fig 3e. The intra-star 176 177 twist angle α increases, approaching 90° below 50 K (Fig. 3d), while the local inter-star angle β reaches 150° as 178 SoDs align and form fully commensurate order (Fig. 3f) and the undistorted phase diminishes, consistent with 179 vanishing domain walls (Fig 3c). Simultaneously, well defined inter-bilayer correlations develop (Fig 4 b,d). The 180 most dramatic changes are observed at ~13.1 Å where PDF intensity redistributes and sharpens, possibly 181 indicating an offset of pairs of SoDs, going from one bilayer to another, by one $P\overline{3}m$ lattice spacing along both 182 planar lattice vectors, Fig 4e. Notably, there are 6 choices of such offsets, consistent with 13x stacking period 183 and helical stacking arrangement suggested in some works⁴⁴. Further qualitative changes are seen below 50 K in the PDF peak which includes the nearest neighbor intra-bilayer stacking peak just below 6 Å, whose intensity 184 185 distinctly and abruptly shifts to lower distance (Figs. 4a,c). In contrast, the inter-bilayer peak at ~13.1 Å sharpens, but does not shift (Figs. 4b,d). This could imply enhanced intra-bilayer coupling, with neighboring bilayers 186 187 presumably adjusting to preserve the inter-bilayer spacings.³⁶

188 Symmetry of the order parameter. On the basis of a symmetry analysis of the observed PDF displacements we can identify the symmetry of order parameters which transform as the A_{2u} and E_u irreducible representations 189 in the parent P $\overline{3}$ m structure space group. The displacements are described by two modes with wavevectors of 190 the form $q_1 = (a, b, 0)$ and $q_2 = (-b, a + b, 0)$ at 60° to q_1 . In the C phase, $q_1 = \frac{3}{13}a^* + \frac{1}{13}b^*$, and $q_2 = \frac{3}{13}a^* + \frac{1}{13}a^* + \frac{1}{13}a^*$. 191 $-\frac{1}{13}a^* + \frac{4}{13}b^*$, where a^* and b^* are the reciprocal lattice vectors of the undistorted lattice. It is possible to 192 193 decompose the atomic displacements of the Ta atoms at any temperature in terms of the independent 194 amplitudes of the A_{2u} and E_u modes described by q_1 and $q_1 + q_2$. The displacements at 15 K are shown in the 195 insert to Figure 3f. In the NC phase at room temperature⁴⁴, $a_{NC} = 0.2448(2)$ and $b_{NC} = 0.0681(2)$, which is close to 196 the C values ($a_c = 3/13$ and $b_c = 1/13$), while in the IC phase, $a_{IC} = 0.283(2)$, and b = 0, where a_{IC} is close to 197 2/7=0.2857. Note that our PDF analysis does not reveal any information about long range order, and q_1 and q_2 198 should not be interpreted to imply its existence. Rather, they describe the symmetry of the lattice distortions 199 associated with the twists and displacements of the Ta atoms in the distorted SoD polaron structures. (See 200 Supplementary Fig. 4).

201 DISCUSSION

The local structure data in 1T-TaS₂ reveals symmetry-specific local structural fingerprints of polaron formation whose ordering evolves with temperature as shown schematically in Fig. 4f. Surprisingly, the polaron fingerprints are already identifiable in the polymorphic 6R-TaS₂ regime (where individual 1T-monolayers appear between 1H layers), in the metallic phase (> 600 K), then successively through the IC, NC and C ordering transitions, and 206 eventually in the non-QSL-like regime below 50 K, where – remarkably – SoD symmetry is restored in a layer-207 dimerized undistorted in-plane SoD structure. An important fundamental question arises in the context of 208 conventional CDW viewpoint regarding the lattice distortions in the high-temperature M phase: Should the 209 deformations be discussed in terms of a uniform reduced modulation amplitude in the IC-like phase, or in terms 210 of dynamical polaron fluctuations? Figure 1d indicates the range of the structural correlations. The black trace 211 shows a difference across the IC-M transition, while the red trace is a difference corresponding to two PDFs in 212 the same (M) phase separated by the same ΔT . This comparison shows that dominant changes across IC-M occur 213 at r > 5 Å, with smaller changes for r < 5 Å (Supplementary Fig. 5). This implies that at the M to IC transition 214 both the number of distorted sites and the extent of IC spatial correlations grow. The M phase thus looks like a 215 sparse polaron gas, whereas the IC phase is a sparse polaron crystal (incommensurate with the lattice). We can 216 now consider the three possible scenarios for the IC melting transition based on the PDF data: In the first one, 217 in the M phase, the long-range IC-CDW becomes dynamic, with amplitude and phase fluctuations on a timescale 218 faster than the lattice can respond. In the second scenario, all relevant charges become fully itinerant on all 219 length scales in the M phase without lattice distortions. In the third, polaronic, scenario the IC-CDW breaks up 220 in the M regime but the lattice follows the localized charge fluctuations, which is equivalent to thermally 221 activated polaron hopping. In all three scenarios, we would see a disappearance of the reflections at the IC 222 modulation wave vector on going from IC to M in Bragg reflections, with the third scenario leaving some diffuse 223 scattering in the M phase. In the PDF data, the first and second scenario would result in no characteristic 224 polaronic distortions in the M phase. The fact that we do see them implies that only the polaron fluctuation 225 scenario is relevant in 1T-TaS₂. The persistence of polarons in the polymorphic regime (Fig. 2 b,c) and the change 226 from 20 wt% to 10 wt% on the transition from a uniform 1T to the 6R multilayer structure confirms this notion 227 and implies that polarons exist only in individual 1T monolayers. These may be expected to appear also in free-228 standing monolayers or monolayers on substrates, provided that strain and interaction with the substrate do 229 not interfere.

230 Magnetic ordering. It is interesting to consider the symmetry of the magnetic polaron structures that are 231 compatible with the two possible in-plane magnetic space groups $P\overline{3}$ (#147.13) and $P\overline{3}$ ' (147.15). Two 232 (ferro)magnetic P3m subgroups can be derived from the analysis of the symmetric irreducible representation at 233 the Γ point of the trigonal Brillouin zone of the 1T stack. For P $\overline{3}m$, inversion does not flip the magnetic moment, 234 allowing a magnetic moment component along c for Ta1 and no constraints for the magnetic moments of the 235 Ta2 and Ta3 atoms. In contrast, for the $P\overline{3}$ 'm magnetic space group, inversion reverses time and thus flips the 236 magnetic moment, the magnetic moments of the Ta atoms of the same kind are not flipped by the symmetry 237 operators. By choosing different components for Ta2 and Ta3 magnetic moments it is possible to create different 238 motifs maintaining the same symmetry. Figs. 4 g,h show the motifs compatible with the symmetry operations 239 of the two magnetic groups for in-plane chiral and non-chiral arrangements of magnetic moments respectively. 240 An in-plane QSL-like state is consistent with fluctuating in-plane moments, and c-axis antiparallel moments in 241 the $P\overline{3}'$ magnetic group, consistent with dimerized stacking of CDW orders in the C state.

242 A remarkable feature of the data is that local in-plane mirror symmetries appear to be restored below 50 K, with 243 $\alpha \simeq 120^{\circ}$ and $\beta \simeq 150^{\circ}$. The data below 50 K are strongly suggestive of dimerization along the c axis, consistent 244 with inter-plane spin singlet formation and spin gap formation, that could explain the apparent cross-over from 245 a $\sim T^2$ QSL-like spin relaxation above 50 K to a gapped, exponential T dependence below this temperature². A 246 plausible origin of the apparent symmetry restoration at low temperature comes from the disappearance of 247 domain walls at low temperature. In-plane domain walls have been observed to persist well below the NC-C 248 transition temperature, which topologically prevent interlayer stacking and spin dimerization. As long as domain 249 walls exist, un-dimerized layers support an intra-layer QSL composed of spin structures shown in Figs. 4g and h, 250 while regions with dimerized lattice layers are expected to exhibit inter-layer spin pairing³. The restoration of 251 SoD symmetry at lowest temperatures may thus be considered to be associated with the disappearance of DWs. 252 A slightly different situation occurs in the metastable 'hidden' domain state⁶⁵, where a QSL is allowed to form 253 within domains whenever inter-layer stacking is indirect.

254 The polaron crystallization paradigm⁶⁶ emerging on the basis of these data has the benefit of providing a 255 common framework for the understanding the polaronic physics in both the equilibrium and metastable phases 256 of 1T-TaS₂ and numerous related materials that display similar charge and spin ordering²³. The local structure 257 symmetry analysis is particularly relevant for revealing the origin of the still poorly understood spin polaron 258 structure and reconciles the observation of QSL-like behavior with theoretical band structure modelling, which 259 commonly suggests a spin-paired dimerized ground state in a perfect crystal³. Finally, we note that it was shown 260 theoretically that Wigner-crystal ordering is not limited to systems with a half-filled electronic band, but may be also present at other fillings^{21,67}, which suggest that the crystallization phenomenon reported here may be more 261 262 general than previously thought.

263 METHODS

Crystal growth and analysis. The 1T-TaS₂ samples were grown using iodine vapor transport reaction, grown at 850 °*C*, and quenched to room temperature from the growth temperature. Single crystal XRD shows a pure single 1T phase is retained after the quench, with Ta:S composition determined by EDS to be 33: $66 \pm 1 at \%$.

267 Synchrotron X-ray atomic pair distribution function (PDF) measurements: Temperature dependent X-ray total 268 scattering data were collected at 28-ID-1 beamline of the National Synchrotron Light Source II (NSLS II) at 269 Brookhaven National Laboratory. Finely ground powders of 1T-TaS₂ were sealed in 1mm (outer diameter) 270 polyimide capillary (referred to as experiment 1 or exp 1) and 1.5 mm (outer diameter) quartz capillary 271 (experiment 2 or exp 2) within a glovebox under light vacuum. Measurements were carried out in capillary 272 transmission geometry using a 2D PerkinElmer amorphous silicon area detector (2048 × 2048 pixels with 200 273 μ m² pixel size) placed ~204 mm downstream of the sample. The setup utilized a monochromatic X-ray beam 274 with 74.5 keV energy (λ = 0.1665 Å). Sample temperature control was achieved using a Cryo Industries of America 275 cryostat (exp 1, 15 K \leq T \leq 500 K) and a FMB Oxford Hot Air Blower model GSB1300 (exp 2, 300 K \leq T \leq 915 K). 276 Data for each experiment were collected in 5 K steps using 5 K/minute temperature ramp, 2 minutes 277 thermalization, and 2 minutes data collection at each temperature. In exp1 data in temperature range 15 K \leq T 278 \leq 300 K were collected on cooling, while these in temperature range 300 K \leq T \leq 500 K were collected on 279 warming and cooling. In exp 2 data were collected on warming.

280 PDF data processing and analysis: Calibrations of the experimental geometry, momentum transfer range, 281 and detector orientation were carried out by utilizing nickel standard measurements performed under the same 282 conditions. Appropriate masking of the beam-stop shadow, inactive and outlier pixels, and subsequent 283 azimuthal integration of the 2D images to obtain 1D diffraction patterns of intensity versus Q data were done using pyFAI software package^{68,69} Standardized corrections to the data for experimental effects to obtain the 284 reduced total scattering structure function, F(Q), and the subsequent sine Fourier transforms to obtain 285 286 experimental PDFs, G(r), with $Q_{max}=25 \text{ Å}^{-1}$ (exp 1) and $Q_{max}=20 \text{ Å}^{-1}$ (exp 2) were carried out using the PDFgetX3 287 program within the xPDFsuite software package⁷⁰ The PDF analysis was carried out using the PDFgui⁷¹ modeling 288 platform. The PDF, derived from powder diffraction-based Bragg and diffuse scattering data collected over a 289 broad range of momentum transfer, describes direction-averaged distribution of atom pairs in a material as a 290 function of interatomic distance, r, and provides structural information across different length scales. 291 Uncertainties in reported model-derived parameters were estimated from their maximum variations observed 292 by changing either the r-range of modelling (Fig. 1b, Fig. 3a-f) or integration (Fig. 2b, inset) by 10%. A double of 293 the maximal values of such uncertainties was adopted as the error bars shown. Due to large X-ray scattering 294 contrast (Z(Ta)=73, Z(S)=16) the dominant contribution to total PDF (Fig. 1e) originates from Ta-Ta pairs (red 295 trace), followed by the Ta-S pairs' contribution (blue trace). The S-S (green trace) contribution is about 20 times 296 weaker than the Ta-Ta contribution and 4 times weaker than that of Ta-S pairs.

Estimate of distorted fraction and structure modeling in polymorphic regime: By using a calculated PDF based
 on undistorted P3m model as a reference, we form difference PDFs for all T>600 K data. Such differences are

299 then integrated over a 2.6-4 Å range (highlighted region on abscissa in Fig. 2b) and normalized to the 600 K value 300 to quantify the relative change with temperature of the fraction of distorted sites across the polymorphic 301 transformations, as shown in inset to Fig. 2b (for illustration see Supplementary Fig. 1). Above 730 K the short-302 range PDF data can be explained by a 6R-TaS₂-like model (R3m symmetry) featuring alternating 1T and 1H layers 303 in equal abundance, Fig. 2c (bottom). Although the exact stacking and long-range character are likely more 304 complex, as the model does not fully explain the data over longer length-scales, local analysis is rather robust. 305 Importantly, the 6R-TaS₂ model features undistorted 1T & 1H layers, hence not accounting for local distortions 306 responsible for nontrivial nearest neighbor Ta-Ta distance distribution.

307 Local structure model with P3 symmetry: To map out temperature evolution of local interatomic distances, first 308 data fits were carried out over 1 nm range and utilized a simple single S-Ta-S layer 1T-TaS₂ model conforming to 309 P3 symmetry. The model includes 19 independent structure parameters, constrained by symmetry to describe 310 fractional coordinates of Ta (6 parameters) and S (13 parameters) atoms in the unit cell, two lattice parameters, 311 two thermal parameters, and an overall scale factor. This explains the data features in the C state very well. The 312 refinement range was established as a compromise between the model over-parametrization (data window too 313 narrow) and the model deficiency (data window too wide) regimes. The 1 nm fitting range encompasses intra-314 bilayer correlations but excludes inter-bilayer correlations, which were subsequently explored in a model-315 independent manner as described below. (See Supplementary Table 1, Supplementary Figures 6 and 7, and 316 Section 2 for additional model details.)

317 **Multiphase treatment:** The fits using only the $P\overline{3}$ phase in the NC regime were noticeably worse compared to 318 these in the C state, with observable increase of fit residual, as shown in inset to Fig. 3c. This is consistent with 319 the presence of discommensuration domain walls and lower SoD density in the NC state whose atomic structure 320 is less distorted and not accounted for in our simple distorted $P\overline{3}$ model. We approximated domain wall 321 contribution by adding an undistorted $P\bar{3}m$ structure as a secondary (minority) phase to account for local 322 environment of the discommensuration regions. We utilized this approach to fit the PDF data in the IC and M 323 regimes as well, where the distorted and undistorted phases swap roles and the distorted phase becomes a 324 minority phase.

325 Assignment and consideration of inter-layer PDF peaks: The inter-layer PDF peaks considered in this study were 326 assigned based on undistorted P3m model as reported for 300 K and its supercell variants. In this, interlayer 327 spacings along the stacking direction are defined by multiples of c-axis lattice parameter, including ~5.9 Å which 328 corresponds to intra-bilayer separation. The \sim 13.1 Å peak in undistorted P3m structure corresponds to a well-329 defined off-axis Ta-Ta contribution connecting two Ta atoms that are 2 layers apart. These Ta atoms belong to 330 next nearest neighbor Ta-layers and are offset within planes away from the stacking axis. In the high temperature 331 regime, if structural distortions are incoherent, this peak in the data would be smeared, as is observed. As order 332 establishes, re-appearance (with respect to putative $P\overline{3}m$ structure that never materializes locally) of a sharp 333 signal at ~13.1 Å implies re-emergence of structural coherence along this interatomic vector. The signal strength 334 likely implies alignment of SoDs belonging to neighboring bilayers along this direction.

335 DATA AVAILABILITY

All of the data supporting the conclusions are available within the article and the Supplementary Information.

337 Additional data are available from the corresponding authors upon request.

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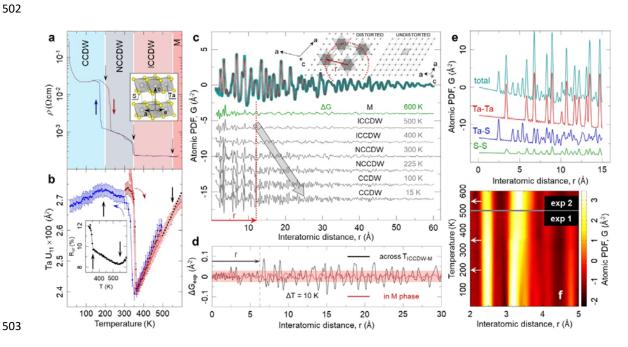
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495 AUTHOR CONTRIBUTIONS

- 496 EB and MA designed and conducted the X-ray experiments, EB performed PDF analysis, GB and SC performed
- 497 group theoretical analysis. DM and EB conceptually proposed the experiments. EB and DM wrote the paper. PS498 synthesized and characterized the crystals.

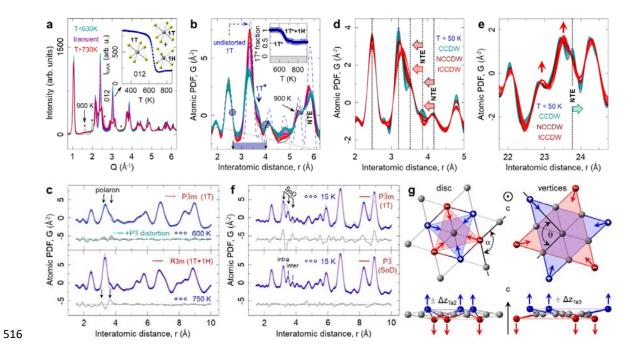
499 COMPETING INTERESTS

500 The authors declare no competing interests in this work.

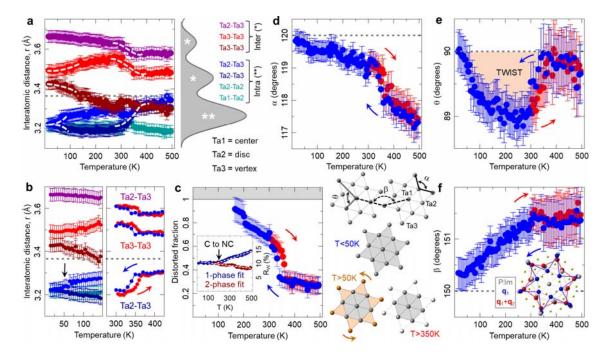


504 Fig. 1 Electronic transitions & lattice symmetry breaking in 1T-TaS₂. (a) Temperature dependent resistivity, 505 adopted from Sipos et al.⁹, highlighting electronic phases in 1T-TaS₂. The insert shows the undistorted P $\overline{3}m$ 506 lattice structure. (b) Temperature evolution of in-plane ADP of Ta from P3m model fit to PDF data over 20-60 Å 507 range. Red color indicates heating, blue color indicates cooling. Sloping black dotted line is a reference. Inset: Fit 508 residual, Rw, implicates IC-M and IC-NC transitions. (c) P3m model fit to 600 K data. Green difference trace (offset 509 for clarity) reveals short range deviations. Stack of difference curves for the same model against data at different 510 temperatures (as indicated) is shown underneath. Red arrow indicates a length-scale coarsely corresponding to 511 the SoD polaron and the in-plane nearest neighbor inter-star center-to-center distance (inset). (d) Comparison 512 of difference PDF between experimental data that are 10 K apart within metallic phase (T > 550 K) and across 513 the IC-M transition, implicating existence of local distortions in the M phase over at least a radius r. (e) Calculated 514 total PDF and partial pair-contributions in the P $\overline{3}$ m structure. (f) False color plot of PDF intensity versus 515 temperature over a short *r*-range.

501



517 Fig. 2 Temperature progression of atomic structure in 1T-TaS₂. (a) Stack of diffraction patterns revealing a 518 polymporphic transformation on warming at ~630 K. Inset: intensity collapse of the (012) P3m Bragg reflection 519 during the restructuring from pure 1T to a hybrid 1T & 1H coordination. Further restructuring above 880 K results 520 in additional Bragg reflections, marked by asterisk. (b) Corresponding PDFs over a narrow r-range. Data feature 521 isosbestic points (circled) - nodes of temperature-invariant PDF intensity. Label 1T* indicates presence of local 522 distortions in the 1T-type layers. Vertical blue arrow (1T*) marks polaronic distortion intensity. Inset: Change of 523 fraction of distorted 1T* environments, relative to 600 K, estimated over the shaded region on the abscissa. (c) 524 Polaron signature, seen in the M phase (top), persists in the restacked polymorph structure (bottom), and is 525 associated with the 1T-type layers. Adding distortions, described in (g), as nanoscale inclusions results in 526 improved fit (cyan trace). In (d) and (e) a stack of PDF data on different length scales is shown over the 15 K -527 500 K range, color-coded according to the electronic phase they belong to. At high temperature longer length-528 scale peaks sharpen (indicated by vertical arrows) due to apparent average symmetry increase, despite expected 529 broadening due to ADP increase at elevated temperature and in contrast to the shorter length-scale behavior. 530 Features exhibiting negative thermal expansion (NTE) are indicated by block arrows. (f) At 15 K clear distortions 531 associated with the star of David (SoD) are seen in the local structure, incompatible with the $P\overline{3}m$ symmetry (top) but explained well (bottom) by a simple $P\overline{3}$ broken symmetry model, depicted in (g) with focus on Ta atoms. 532 533 Colored arrows show the Ta displacements. Out-of-plane degrees of freedom (exaggerated in the illustration for 534 clarity) provide for puckering distortions.



535

536 Fig. 3 Local structure quantification from distorted local SoD model. (a) Local Ta-Ta distances from a $P\overline{3}$ 537 model fits to PDF data, obtained on cooling, over 1 nm length-scale in 15-500 K range. For reference, dashed 538 horizontal gray line marks the undistorted P3m value at 300 K. Intra- and inter-star distances are color coded as 539 indicated in the legend. (b) Closeup of C phase behavior (left) and hysteretic response across the IC-NC transition 540 (right). (c) In NC and IC phases secondary undistorted $P\overline{3}m$ phase was necessary to reduce fit residual, R_w (inset). 541 Distorted phase fraction, displaying thermal hysteresis consistent with one seen in resistivity, is calculated from 542 atomic weight-based phase content obtained from the fits. (d)-(f) show selected intra-SoD and inter-SoD bond 543 angles, as indicated in the sketch. In the IC phase the SoD motif is heavily distorted, with local distortions 544 resembling discs. Upon entering the NC phase, the SoD distortion becomes better differentiated, with stars 545 exhibit twists, and with interatomic angles beginning to evolve towards ideal values. In the C phase twists start 546 to diminish. Below ~50 K a sharp regularization of stars is observed, with intra-star Ta-Ta distances becoming 547 equal, intra-star angles approaching ideal values, and no twisting. Colored arrows in (b)-(f) indicate data 548 collection on warming (red) and on cooling (blue) cycles. Inset to (f): displacements of the Ta atoms for the q_1 549 (red) and $\mathbf{q}_1 + \mathbf{q}_2$ (blue) modes from symmetry analysis at 15 K, compared to the parent P $\overline{3}$ m phase (gray).

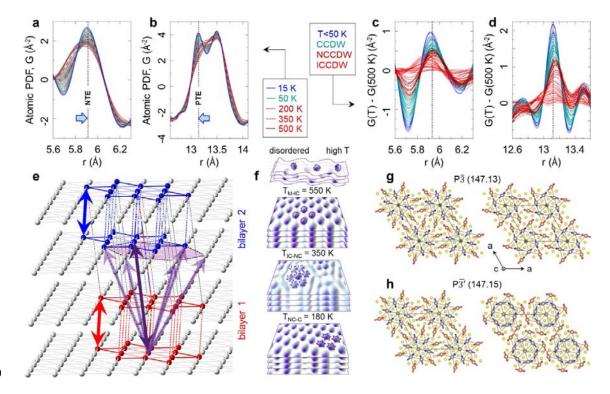




Fig. 4 Model independent considerations of stacking correlations. (a) Temperature stack of experimental 551 552 PDFs around ~5.9 Å peak containing *intra*-bilayer correlations of SoDs, indicated in (e) by blue and red double 553 arrows. (b) Temperature stack of PDFs around ~13.1 Å peak depicting inter-bilayer correlations of SoDs, 554 indicated in (e) by purple double arrows. The contribution assignment is made based on features in P $\overline{3}$ m model, 555 Fig. 1e. There, the 5.9 Å feature is the c-axis lattice repeat distance involving strong signal from the nearest 556 neighbor interlayer Ta-Ta correlations, whereas the 13.1 Å feature is well isolated and dominated by the next 557 nearest neighbor off-axis interlayer Ta-Ta correlations. In (c) and (d) temperature stacks of differential PDFs are 558 shown for data in (a) and (b), respectively, with differentials calculated using 500 K reference. Vertical dashed 559 lines mark positions of apparent maxima in the C phase. (e) Sketch of two bilayers highlighting correlations 560 discussed in text. Inter-bilayer SoDs are separated by ~13.1 Å at low temperature, a peak forming on cooling 561 from a broad distribution at high temperature. This separation results from a SoD-shift by approximately one 562 $P\overline{3}m$ lattice spacing in each $P\overline{3}m$ lattice direction. Note 6 possible choices of relative positioning of nearest 563 bilayers, indicated by the purple shaded hexagon. (f) Schematics of distortions across different electronic phases 564 in $1T-TaS_2$. (g, h) magnetic moments consistent with the distortions of the SoDs as determined by the PDF 565 analysis for two different settings. The arrows at the center Ta1 atom pointing along the z axis are shown in 566 green (only in $P\overline{3}m$).

567

Supplementary Information

Crystallization of polarons through charge and spin ordering transitions in $1T\text{-}TaS_2$

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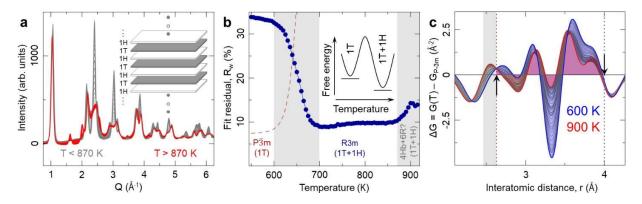
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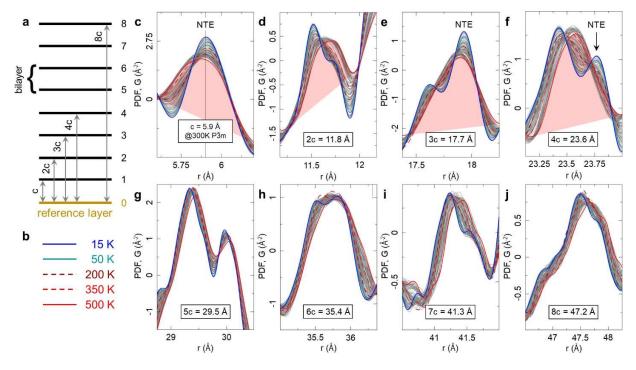
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1. SUPPLEMENTARY FIGURES AND TABLES



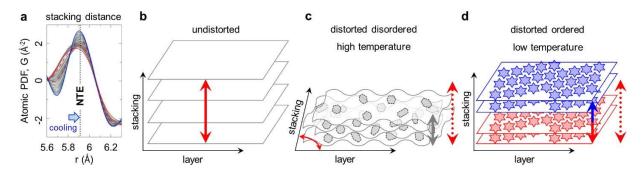
Supplementary Figure 1. Polymorphic transformations in 1T-TaS₂ on warming in the M phase.

a Stack of diffraction patterns revealing two polymorphic transformations on warming, T > 600 K. Red stack data, T > 870 K, reveal additional transformation to that focused on in the main text. This additional restructuring likely involves a distorted mixture of 4Hb (P63/mmc) and 6R (R3m) polymorphs with significant stacking faults. Low resolution of our diffraction data did not allow unambiguous characterization. Tentative PDF assessment suggested a complex mixture of alternating 1T and 1H layers (inset). **b** Transformation temperatures charted by fit residuals of P $\overline{3}$ m and R3m models, as indicated in the panel. Inset: sketch of the free energy of TaS₂, depicting metastable state (1T) and more stable configurations (1T+1H) achieved as temperature is elevated. **c** Stack of narrow *r*-range difference PDFs, Δ G, obtained by subtracting simulated PDF from a literature reference based on undistorted P $\overline{3}$ m at 300 K, from experimental PDFs (lower r-resolution, Q_{max}=20 Å⁻¹) collected at T > 600 K. The difference PDF is acquired for all M-phase data. To estimate the change in fraction across the polymorphic transformations, the difference PDF, Δ G, was integrated between 2.6 – 4 Å (isosbestic points encompassing PDF feature of interest, between vertical arrows) for each temperature, and the final integral was normalized to the value obtained at 600 K. The uncertainties in the distorted fraction were estimated by extending the integration range over the grey window (nearest low-*r* zero crossings) and repeating the process. A double of the maximal value of such uncertainty was adopted as the error bars shown in the inset of Fig. 2b of the main text.



Supplementary Figure 2. Evolution of stacking correlations in 1T-TaS₂.

a Idealized stacking of Ta-layers, with interlayer spacings as indicated, assigned based on $P\overline{3}m$ structure, and defined with respect to a reference layer. **b** Legend of color-coded temperatures of selected *r*-range PDF data shown in panels **c**-**j**. In these panels the r-ranges were selected to encompass features associated with the multiples of the $P\overline{3}m$ c-axis lattice parameter reported in literature for 300 K. While the PDF traces corresponding to the transitions to equilibrium electronic phases are colour coded, the other data in the stacks are shown in grey. The correlations up to 8c are shown. Profiles at 500 K are highlighted in red up to 4c range to emphasize the broadness of the high temperature signal which in part originates from interlayer disorder. Features with apparent negative thermal expansion characteristics are labelled as NTE.

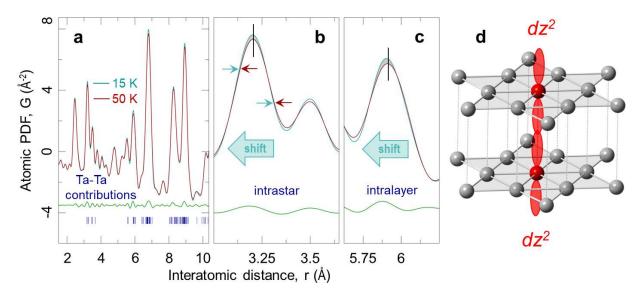


Supplementary Figure 3. Possible origin of the c-axis NTE jumps.

a Stack of PDF data around the nearest neighbour layer stacking distance at ~5.9 Å, as presented in Fig. 4a of the Main Text, and in Suppl. Fig. 3c. Dashed vertical line marks c-axis lattice parameter in undistorted $P\overline{3}m$ structure. Coloured traces mark 500 K (solid red line), 350 K (dashed red line), 200 K (dashed dark red line), 50 K (solid cyan line) and 15 K (solid blue line). **b.** In $P\overline{3}m$ structure layers are equidistant, as sketched. **c.** In distorted heavily disordered structure, such as that observed at higher temperatures, puckering distortions result in heavily distorted layers, resulting in interlayer stacking disorder and observed broad distribution of nearest neighbour interlayer distances, such as seen in **a.** In proposed NTE model this results in layer-shears and monoclinic compressions yielding shorter-than-expected average c-lattice parameter. **d.** As interlayer orders set in on cooling, the puff-pastry-like stacking disorder gets lifted, and the c-axis expands resulting in observed NTE jumps, as described in text.

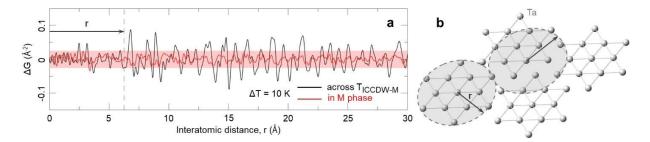
Origin of c-axis negative thermal expansion (NTE) jumps

Dilatometry measurements in 1T-TaS2 [1] revealed anomalous NTE jumps in c-axis parameter, most significantly at the NC-C transition, but also at the IC-NC transition, origin of which has not been understood to date. Our data also reveal NTE behavior of certain PDF features, as marked in Figs. 2b,d,e and 4a of the Main Text, as well as in Supplementary Figure 3, and offer a simple explanation for this phenomenon. Based on these observations we propose a disorder-based rationale for NTE jumps. In putative undistorted $P\overline{3}m$ crystal structure 1T layers stacked along c-axis are equidistant, Suppl. Fig. 4b. However, as our PDF analysis demonstrates, 1T-TaS2 is locally distorted at all temperatures at which this metastable polymorph exists, including the highest temperature in the M phase, resulting in 1T*-TaS2. There, the distortions involve local puckering of Ta layers, such as schematically shown the top panel of Fig. 4f in the Main Text as well as in Supp. Fig. 4c. As a consequence, disordered layers exhibit a range of interlayer distances and concomitant sheer displacements due to Coulomb repulsion of polarons in neighbouring layers, evident from broad interlayer distance distributions that are, importantly, centered at nominal distance shorter than simple integer multiples of undistorted $P\overline{3}m$ c-axis lattice parameter (Main Text Fig. 4a, c at high temperature, and also Supplementary Fig. 3c and 4a) resulting in, on average, closer packing of the layers. Due to this, the experimentally measured reference c-axis lattice parameter at high temperature is anomalously short, resembling puff-pastry configuration as depicted in Fig. 4 (top) and Supp. Fig. 4c. As polaron gas condenses into a Wigner crystal, and the distortions become long range ordered and stacked, aided by Coulomb repulsion effects, the ordered layers space away, effectively removing monoclinic shears, resulting in increase of the average interlayer spacing (Main Text Fig. 4 a,c at low temperature), as sketched in Supp. Fig. 4d, leading to apparent NTE jumps observed in dilatometry.





a An overplot of experimental PDFs of 15 K (cyan trace) and 50 K (red trace), with corresponding differential (green trace) offset for clarity. The Ta-Ta contributions per $P\overline{3}m$ model used in this work are shown as blue ticks. **b** The rregion sensitive to intrastar correlations. Apart from sharpening, expected upon temperature reduction, centroid of the relevant peak also shifts to the left, consistent with SoD regularization revealed by modelling show in the Main Text. c The r-region sensitive to the nearest neighbour intra-bilayer SoD correlations. The peak also exhibits observable shifting, consistent with enhanced inter-bilayer binding. Vertical black line in b and c marks the centroid of feature of interest at 50 K. One possible origin of this behaviour, as discussed in the Main Text, is a formation of spin-singlet dimers associated with the stray spins, presumably from the dz^2 manifolds, in two adjacent SoDs within the bilayers. Notably, the amount of change in PDF signal seen in c is quite large. It is unlikely than only one Ta-Ta contact per SoD (1:13) contracts, implying that the mechanism is more complex than simple pairing of spins in dz^2 orbitals portrayed in d. Importantly, the features in the differential in a are well structured, stronger under some PDF peaks and weaker under others. Given that the differential is of two data sets, it is unlikely that they originate from artifacts such as termination ripples or noise in the data. Even if present, such effects would effectively cancel out in the subtraction, leaving only signature of subtle structural differences. We further note that as part of the experiment we carried out a counting statistics test to assess the signal to noise ratio. In this, data statistics was found to be satisfactory after counting for 10 seconds. The actual counting time was 120 seconds, an order of magnitude longer.



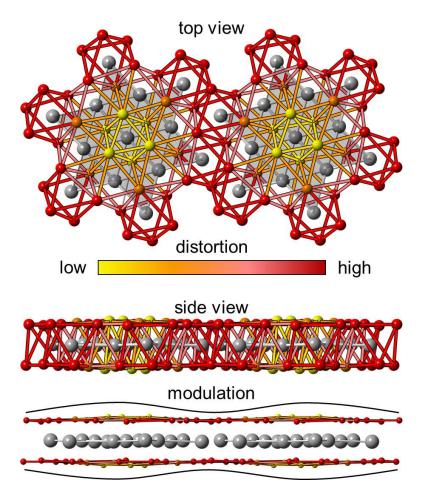
Supplementary Figure 5. Local structure changes across M-IC transition in 1T-TaS₂.

a Comparison of difference PDF between data that are 10 K apart within metallic phase (T > 550 K) and across the IC-M transition, implicating existence of local distortions in the M phase over at least a radius r, as sketched in **b**. Note that these difficult-to-model distortions do not reflect star or David in the M phase, but heavily puckered hexagonal discs as indicated by modelling presented in the Main Text. Enhancement of difference signal on the length-scale r > 5 Å implies growth of structural correlations in the IC regime.

				Tantalum	x	у	z
Q				1	-	-	-
					p1	p2	р3
)	3	1-p2	р1-р2	р3
	5			4	1-p1+p2	1-p1	р3
				5	1-p1	1-p2	-p3
b	10 13	Leto		6	p2	1-p1+p2	-p3
	$\Delta \Lambda$	9 8		7	р1-р2	p1	-p3
\odot	4	10		8	p4	p5	p6
С		\rightarrow 2		9	1-p5	p4-p5	p6
۵.	XM	a 🏹		10	1-p4+p5	1-p4	p6
	The The				1-p4	1-p5	-p6
ă ă				12	p5	1-p4+p5	-p6
			13	p4-p5	p4	-p6	
				10	P4 P0	PT	-po
Sulfur	x	У	Z	Sulfur	x	у У	z
Sulfur 14	х р7	у p8	z p9				
		-		Sulfur	x	У	z
14	р7	p8	p9	Sulfur 27	x 1-p14	у p13-p14	z p15
14 15	р7 1-р8	р8 1+р7-р8	р9 р9	Sulfur 27 28	x 1-p14 1-p13+p14	у p13-p14 1-p13	z p15 p15
14 15 16	р7 1-р8 -р7+р8	р8 1+р7-р8 1-р7	р9 р9 р9	Sulfur 27 28 29	x 1-p14 1-p13+p14 1-p13	у p13-p14 1-p13 1-p14	z p15 p15 1-p15
14 15 16 17	p7 1-p8 -p7+p8 1-p7	p8 1+p7-p8 1-p7 1-p8	р9 р9 р9 1-р9	Sulfur 27 28 29 30	x 1-p14 1-p13+p14 1-p13 p14	y p13-p14 1-p13 1-p14 1-p13+p14	z p15 p15 1-p15 1-p15
14 15 16 17 18	p7 1-p8 -p7+p8 1-p7 p8	p8 1+p7-p8 1-p7 1-p8 -p7+p8	p9 p9 p9 1-p9 1-p9	Sulfur 27 28 29 30 31	x 1-p14 1-p13+p14 1-p13 p14 p13-p14	y p13-p14 1-p13 1-p14 1-p13+p14 p13	z p15 p15 1-p15 1-p15 1-p15
14 15 16 17 18 19	p7 1-p8 -p7+p8 1-p7 p8 1+p7-p8	p8 1+p7-p8 1-p7 1-p8 -p7+p8 p7	p9 p9 1-p9 1-p9 1-p9 1-p9	Sulfur 27 28 29 30 31 32	x 1-p14 1-p13+p14 1-p13 p14 p13-p14 p16	y p13-p14 1-p13 1-p14 1-p13+p14 p13 p17	z p15 p15 1-p15 1-p15 1-p15 p18
14 15 16 17 18 19 20	p7 1-p8 -p7+p8 1-p7 p8 1+p7-p8 p10	p8 1+p7-p8 1-p7 1-p8 -p7+p8 p7 p11	p9 p9 1-p9 1-p9 1-p9 1-p9 p12	Sulfur 27 28 29 30 31 32 33	x 1-p14 1-p13+p14 1-p13 p14 p13-p14 p16 1-p17	y p13-p14 1-p13 1-p14 1-p13+p14 p13 p17 1+p16-p17	z p15 1-p15 1-p15 1-p15 1-p15 p18 p18
14 15 16 17 18 19 20 21	p7 1-p8 -p7+p8 1-p7 p8 1+p7-p8 p10 1-p10	p8 1+p7-p8 1-p7 1-p8 -p7+p8 p7 p11 p10-p11	p9 p9 1-p9 1-p9 1-p9 1-p9 p12 p12	Sulfur 27 28 29 30 31 32 33 34	x 1-p14 1-p13+p14 1-p13 p14 p13-p14 p16 1-p17 -p16+p17	y p13-p14 1-p13 1-p14 1-p13+p14 p13 p17 1+p16-p17 1-p16	z p15 p15 1-p15 1-p15 1-p15 p18 p18 p18
14 15 16 17 18 19 20 21 21 22	p7 1-p8 -p7+p8 1-p7 p8 1+p7-p8 p10 1-p10 1-p10+p11	p8 1+p7-p8 1-p7 1-p8 -p7+p8 p7 p11 p10-p11 1-p10	p9 p9 1-p9 1-p9 1-p9 p12 p12 p12	Sulfur 27 28 29 30 31 32 33 34 35	x 1-p14 1-p13+p14 1-p13 p14 p13-p14 p16 1-p17 -p16+p17 1-p16	y p13-p14 1-p13 1-p14 1-p13+p14 p13 p17 1+p16-p17 1-p16 1-p17	z p15 1-p15 1-p15 1-p15 p18 p18 p18 p18 1-p18
14 15 16 17 18 19 20 21 21 22 23	p7 1-p8 -p7+p8 1-p7 p8 1+p7-p8 p10 1-p10 1-p10 1-p10+p11 1-p10	p8 1+p7-p8 1-p7 1-p8 -p7+p8 p7 p11 p10-p11 1-p10 1-p11	p9 p9 1-p9 1-p9 1-p9 p12 p12 p12 p12 1-p12	Sulfur 27 28 29 30 31 32 33 34 35 36	x 1-p14 1-p13+p14 1-p13 p14 p13-p14 p16 1-p17 -p16+p17 1-p16 p17	y p13-p14 1-p13 1-p14 1-p13+p14 p13 p17 1+p16-p17 1-p16 1-p17 -p16+p17	z p15 p15 1-p15 1-p15 p18 p18 p18 p18 1-p18 1-p18

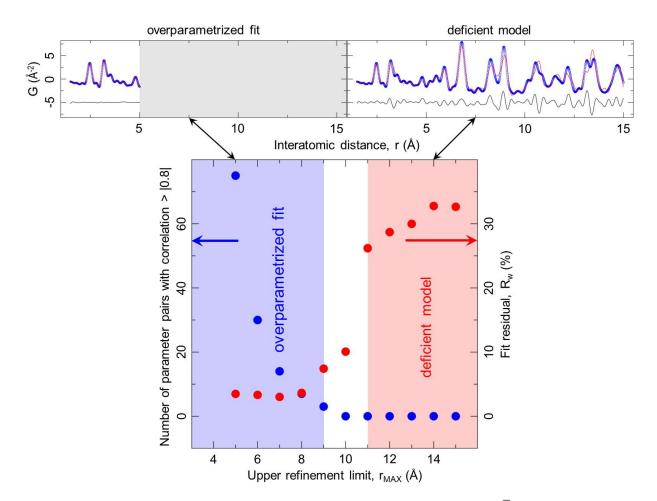
Supplementary Table 1. Details of the single layer $\sqrt{13} \times \sqrt{13}$ supercell P $\overline{3}$ model.

The sketch of the Ta plane, top left, expanded to show four SoDs, with Ta atoms in the model enumerated. Fractional coordinates of Ta atoms are described by 6 independent positional parameters, with symmetry constrains as tabulated in top right. Fractional coordinates of S atoms are described by 13 independent positional parameters, that were symmetry constrained as tabulated (bottom). The model does not account for the c-axis stacking correlations beyond the nearest neighbour layers and does not explain PDF data beyond the length-scale of ~1 nm. Since the fitting range is shorter than the unit cell size, lattice parameters (a=b, c) provide the distance metrics rather than describing translational symmetry. The parameters are labelled as p_i , where i=1-19 represent the index of each symmetry-independent parameter.



Supplementary Figure 6. Features of the P3 structure fit to the 15 K PDF data of 1T-TaS₂ in the C state.

The illustration focuses on two SoD motifs. Tantalum atoms are shown as larger gray spheres, whereas sulphur atoms are depicted as smaller coloured spheres, where colouring reflects their location with respect to the central Ta in SoDs. Moving away from the central Ta the TaS₆ octahedra are progressively more distorted, as portrayed by the colour bar which could be considered as coarse indicator of strain associated with the SoD formation. Although our local model is relatively simple approximant that does not describe the long-range ordering, it does incorporate the degrees of freedom necessary to account for the modulation of sulphur sublattice, as shown in the bottom most panel where the sulphur atoms are represented by very small spheres to emphasize the effect seen in the fit. The solid lines are the guides to the eye and the modulation they depict is out of proportion. Such modulation is expected in the long range ordered structure and is enabled in our approximant model focused on the evolution of local Ta-Ta correlations.



Supplementary Figure 7. Considerations of the fitting range and the applicability of the $P\overline{3}$ approximant model.

In order to establish suitable range for the local structure study, the model was tested against the 15 K dataset. In this, 19 structural parameters, described in Suppl. Fig. 6, were varied, while the lattice parameters, thermal parameters, and the overall scale were kept fixed. The fitting test was conducted over a variable range of data by incrementing the upper limit of refinement, r_{MAX} , by 1 Å, from 5 Å to 15 Å, as shown in the figure. In this process we monitored two quantities: the fit quality captured by the fit residual, Rw, and the number of parameter-pairs whose correlation is larger than |0.8| as reported by the fitting program. In the former, large value of RW indicates inadequacy of the model used. In the latter, parameters exhibiting correlation >|0.8| are typically considered to be unreliable since they are not independent when too narrow a data range is considered. Their report by the fitting platform indicates an over-parametrization of the model or, conversely, insufficient information in the data to constrain the model. In our test, fits of exceptionally good quality were obtained for $r_{MAX} \le 8$ Å, as shown by the RW(T) dependence (red symbols in the bottom panel, right ordinate) and illustrated in the top left panel. However, fits over these ranges result in heavily over-parametrized fits, as indicated by the temperature dependence of the number of correlated parameter pairs (blue symbols in the bottom panel, left ordinate). Increasing the fitting range results in parameters becoming uncorrelated, as sufficient amount of data is added to the fitting. While this initially

helps, with a moderate jump in R_W , for $r_{MAX} \ge 11$ Å the fits begin to deteriorate, as illustrated in the upper right panel. The reason behind this is that wider range of PDF data provides access to significant contributions of the NNN interlayer correlations which are beyond the scope of our present model. We identified $r_{MAX} = 10$ Å as a sweet spot offering a reasonable compromise between the over-parametrization (model too complex for the data range) and underfitting (model inadequacy). The result is a reliable determination of the local structure, which is the focus of the present work.

2. ORDER PARAMETERS DESCRIBING THE SYMMETRY CHANGE LEADING TO THE COMMENSURATE CDW LOW TEMPERATURE PHASE OF 1T-TaS₂.

The analysis of the symmetry of the high temperature structure of $1T-TaS_2$ can help understanding the structural changes of the low temperature daughter structures in terms of the symmetry adapted coordinates belonging to the irreducible representations of the parent space-group symmetry. The physical order parameters describing the change (in this case, the atomic displacements) can be classified as basis functions of the irreducible representations of the parent symmetry group. In the current case, multiple order parameters contribute to the final observed distortion: the individual degrees of freedom provided by an instance of these multi-dimensional order parameters are physically meaningful because they provide a natural description of the distortions of the parent structure.

Lowering the temperature, some of the symmetry elements of the symmetry group of the parent phase are lost (e.g the mirror planes), but the surviving symmetries that describe the daughter phase are a subgroup of the parent phase that is uniquely identified by its combination with the new superlattice basis (and eventual new supercell origin choice). Using a frozen-phonon picture, the structural parameters describing the low temperature daughter structures can be organized as a superposition of polarization vectors containing the cartesian components of the displacements of each one of the independent atoms of the daughter phase.

Revisiting the 1T (single layer) C-CDW: atomic vibrations

In the following, we totally neglect any c-stack rule and consider the symmetry changes occurring in a single layer of the TaS₂ structure. We start from the standard description of the prototype phase that has space group symmetry P-3m1 (#164, D_{3d}^3 , $a_P=b_P\approx 3.34$ Å, $c_P\approx 5.894$ Å, $\gamma=120^\circ$). In this structure, there are two independent atoms with positions described by the coordinates summarized in the table below:

atom	n site	х	У	Z	sym
Ta0	1a	0	0	0	D3d
S0	2d	1/3	2/3	0.22074	C3v

The mechanical representation Γ at the Brillouin zone centre of this prototype structure is reducible and it consists of 2 acoustic and 4 optic mode frequencies as shown by the decomposition into irreducible representations.

$$\Gamma = (A_{2u} \oplus E_u) \oplus (A_{1g} \oplus A_{2u} \oplus E_g \oplus E_u)$$

In particular, the acoustic phonons of this prototype structure belong to the $(A_{2u} \oplus E_u)$ irreducible representations and involve the displacements of the Ta atoms. The A_{2u} irrep describes a vibration of the Ta atom along *z*, while the E_u symmetry adapted displacements describe degenerate vibrations in the *xy* plane.

The low T variants

All the low temperature phases have axes rotated when compared to the original frame of the high temperature prototype structure. The lattice changes produced by rotation around the z-axis (which has A_{2g} representation) can maintain the -3 symmetry operations, but they systematically remove the mirrors and 2-fold axes of the parent group. The low temperature phases also involve superlattice reflections and therefore require the definition of a superlattice cell in real space. The superlattice reflections characteristic of the low temperature variants (IC, NC, and C-CDW phases) involve a condensation of Bragg intensity at some *q*-points in the high temperature Brillouin zone.

In the simplest description of the low temperature variants (incommensurate or commensurate), two additional independent vectors of the high temperature Brillouin zone are required. The low temperature phase (C-CDW) is obtained by the following matrix transformation of the prototype lattice

$$(a \ b \ c) = (a_P \ b_P \ c_P) \begin{pmatrix} 4 & -1 & 0 \\ 1 & 3 & 0 \\ 0 & 0 & 1 \end{pmatrix}$$

In this reference cell, there are 13 replicas of the TaS₂ motif. The atomic positions of the prototype structure can be described in this new metric that has symmetry P-3 (#147, a=b \approx 12.04254 Å, c \approx 5.894 Å, γ =120°) by a new set of independent atomic positions summarized in the table below:

atom	site	x	У	Z
Ta1	1a	0.	0.	0.
Ta2	6g	10/13	1/13	0.00000
Ta3	6g	11/13	5/13	0.00000
S1	6g	-1/3+6/13	1/3-2/13	0.22074
S2	6g	2/3+3/13	1/3-1/13	0.22074
S3	6g	2/3+4/13	1/3+3/13	0.22074
S4	6g	-1/3+7/13	1/3+2/13	0.22074
S5	2d	2/3	1/3	0.22074

This phase corresponds to the lock-in of the components of two vectors q_1 and q_2 ; these reciprocal space vectors have generic components (ab0) and form an angle of 60° to maintain the symmetry of the threefold axis. Their components have the form q_1 = (a, b, 0) and q_2 =(C₋₃₊ q_1)= (-b, a+b,0). At the lock-in, described by the lattice change transformation matrix, $4q_1$ - q_2 =a* and q_1 + $3q_2$ =b* so that q_1 =3/13 a*+1/13 b* and q_2 =-1/13 a*+4/13 b*.

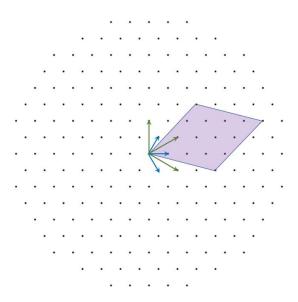
Since there is no component normal to the layer, the q_1 vector defines a star of 6 vectors (including q_2) related by the symmetry operations of the parent space. The same considerations actually apply also to the symmetry analysis of the incommensurate variant studied by Spijkerman et al. In that case, the components a and b of the vectors are incommensurate but not far from their lock-in values:

a_{IC}=0.2448(2) -> a_{C-CDW}=0.2308=3/13 b_{IC}=0.0681(2) -> b_{C-CDW}=0.0769=1/13

As we can see, at the lock-in, the a_{lc} component has decreased and b_{lc} has increased, so the q_1 and q_2 vectors orientations relative to the parent phase have rotated while the temperature was lowered. A

commensurate approximant of the incommensurate phase with P-3 symmetry could be constructed eventually taking a=1/4=0.25 and b=1/16=0.0625, or some other convenient rational approximation of those values.

To calculate the frozen phonons displacements in the C-CDW phase, we have to consider two stars of vectors. The first star is the one already discussed, originating from q_1 (q_2 obviously belongs to this star by construction) and represented by blue arrows in Supplementary Figure 9. The second star of vectors is the one originating from q_1+q_2 with components (2/13 5/13 0) that indexes the 2nd order superlattice reflections of this phase, so that all the 12 extra spots of the reciprocal lattice of the C-CDW phase can be indexed. Each one of these two stars has in principle 12 branches and so the dimension of the Order Parameter for each one of the two stars. The 12 branches of the star of these vectors in P-3m1 have components E(a,b,0), i(-a, -b, 0), -3⁺(-b,a+b,0), 3⁺(b, -a-b, 0), -3⁻ (a+b,-a,0), 3⁻(-a-b, a, 0), m₁₀₀(-a,a+b,0), 2₁₀₀(a, -a-b, 0), m₀₁₀(a+b,-b,0), 2₀₁₀(-a-b, b, 0), 2₁₁₀(b,a,0), and m₁₁₀(-b, -a, 0).



Supplementary Figure 9. Representation of the two stars indexing the superlattice reflections of the C-CDW phase. Elements of the q_1 star are displayed as blue vectors, elements of the q_1+q_2 star are shown as green vectors. The shaded area is the reciprocal cell of the prototype phase.

For the q_1 star we can simply take the spots indexed by the elements (E, -3^+ and -3^-) with explicit components (3/13 1/13 0) (-1/13 4/13 0) (4/13 -3/13 0): therefore, the OP describing the superpositions of the atomic displacements has components (a0a0a0000000). The same choice can be made for the q_1+q_2 star, or explicitly (2/13 5/13 0) (-5/13 7/13 0) (7/13 -2/13 0). The OP describing the transformation from the parent phase to the C-CDW lock-in phase has overall symmetry components

(a0a0a0000000;b0b0b0000000). It is then possible to decompose the atomic displacements of the Ta atoms at any temperature in terms of the amplitudes of the E_u and A_{2u} modes transported at q_1 and at q_1+q_2 .

Symmetry adapted coordinates of Ta displacements at q1 and at q1+q2.

These displacements of the Ta atoms are particularly relevant for the description of the dimerisation of Ta distances that is a characteristic signature of the C-CDW phase.

Eu (q1) 1st component – phases and relative displacement amplitudes along x,y for Ta2 and Ta3 atoms

- Ta1 0.0000 0.0000 0.0000
- Ta2 -0.9708 -1.0000 0.0000
- Ta3 -0.2939 -0.3028 0.0000
- Eu (q₁) 2nd component phases and relative displacement amplitudes along x,y for Ta2 and Ta3 atoms
- Ta1 0.0000 0.0000 0.0000
- Ta2 -0.5222 0.4778 0.0000
- Ta3 -0.1581 0.1447 0.0000

Eu (q1+q2) 1st component – phases and relative displacement amplitudes along x,y for Ta2 and Ta3 atoms

- Ta1 0.0000 0.0000 0.0000
- Ta2 -0.0088 0.2939 0.0000
- Ta3 0.0292 -0.9708 0.0000

Eu (q1+q2) 2nd component – phases and relative displacement amplitudes along x,y for Ta2 and Ta3 atoms

- Ta1 0.0000 0.0000 0.0000
- Ta2 -0.3028 -0.1581 0.0000
- Ta3 1.0000 0.5222 0.0000
- A2u (q1) phases and relative displacement amplitudes along z for Ta2 and Ta3 atoms
- Ta1 0.0000 0.0000 0.0000
- Ta2 0.0000 0.0000 0.3028
- Ta3 0.0000 0.0000 1.0000

A2u (q_1+q_2) – phases and relative displacement amplitudes along z for Ta2 and Ta3 atoms

Ta1 0.0000 0.0000 0.0000

Ta2 0.0000 0.0000 -1.0000

Ta3 0.0000 0.0000 0.3028

This analysis can be also extended to S atoms, but:

- more components are involved, reducing the value of the decomposition procedure
- optic phonons do not seem particularly relevant for this phase transition mechanism

Cartesian components of the displacements of each independent Ta atom in the daughter phase

The supplementary table 2 summarizes the fractional displacement contributions originating from the amplitudes transported at q_1 and q_1+q_2 . These displacements add to the undistorted positions of the Ta2 and Ta3 atoms inherited from the high temperature parent phase. The displacements of the Ta1 atom at the origin of the lattice are fixed by the symmetry element of the daughter phase, so they are not reported in the table. Some trends can be remarked:

- The out-of-plane z component decreases with decreasing T, resulting in puckering of the Ta atoms
- The Eu q₁ component is more important for Ta2 atoms whereas the Eu (q₁+q₂) component is the dominant displacement component for Ta3 atoms.

	undistorted		q1	175K	150K	125K	100K	50K	15k
Ta2_x	0.07692	Eu	Ta2_dx	-0.00957	-0.00957	-0.00898	-0.00903	-0.00838	-0.00779
Ta2_y	0.30769	Eu	Ta2_dy	-0.01577	-0.01576	-0.01536	-0.01541	-0.01548	-0.01529
Ta2_z	0	A2u	Ta2_dz	-0.00219	-0.00210	-0.00151	-0.00162	-0.00162	-0.00169
Ta3_x	0.15385	Eu	Ta3_dx	-0.00188	-0.00188	-0.00193	-0.00193	-0.00215	-0.00227
Та3_у	0.61538	Eu	Ta3_dy	0.00291	0.00290	0.00272	0.00273	0.00254	0.00236
Ta3_z	0	A2u	Ta3_dz	-0.00727	-0.00695	-0.00498	-0.00535	-0.00536	-0.00560
			q1+q2	175K	150K	125K	100K	50K	15K
		Eu	Ta2_dx	0.00311	0.00320	0.00341	0.00359	0.00392	0.00410

Eu	Ta2_dy	0.00405	0.00404	0.00443	0.00445	0.00446	0.00446
A2u	Ta2_dz	0.00894	0.00877	0.00797	0.00755	0.00722	0.00731
Eu	Ta3_dx	-0.00313	-0.00278	-0.00337	-0.00285	-0.00178	-0.00118
Eu	Ta3_dy	0.01027	0.01056	0.01125	0.01185	0.01295	0.01355
A2u	Ta3_dz	-0.00271	-0.00265	-0.00241	-0.00229	-0.00218	-0.00221

Supplementary table 2. Fractional displacement contributions originating from the amplitudes transported at q_1 and q_1+q_2 .

[1] O. Sezerman, A. M. Simpson, and M. H. Jericho, Solid State Commun 36, 737 (1980).