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Protection of Ising spin-orbit coupling in bulk misfit superconductors

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Low-dimensional materials have remarkable properties that are distinct from their bulk counterparts. A paradigmatic example is Ising superconductivity that occurs in monolayer materials such as NbSe₂ which show a strong violation of the Pauli limit. In monolayers, this occurs due to a combination of broken inversion symmetry and spin-orbit coupling that locks the spins of the electrons out-of-plane. Bulk NbSe₂ is centrosymmetric and is therefore not an Ising superconductor. We show that bulk misfit compound superconductors, (LaSe)_{1.14}(NbSe₂) and (LaSe)_{1.14}(NbSe₂)₂, comprised of monolayers and bilayers of NbSe₂, exhibit unexpected Ising protection with a Pauli-limit violation

comparable to monolayer NbSe₂, despite formally having inversion symmetry. We study these misfit compounds using complementary experimental methods in combination with first-principles calculations. We propose theoretical mechanisms of how the Ising protection can survive in bulk materials. We show how some of these mechanisms operate in these bulk compounds due to a concerted effect of charge-transfer, defects, reduction of interlayer hopping, and stacking. This highlights how Ising superconductivity can, unexpectedly, arise in bulk materials, and possibly enable the design of bulk superconductors that are resilient to magnetic fields.

In superconductors, an external magnetic field can break the Cooper pairs via two mechanisms. The first is orbital pair breaking due to the Lorentz force exerted by this field. This does not occur in two-dimensional (2D) superconductors if the magnetic field is in-plane. The second mechanism is spin pair breaking due to Zeeman shifts of opposite signs on the antiparallel spins in a spin-singlet Cooper pair. The minimum field to break pairs is the Pauli paramagnetic limit $B_P^{1,2}$. It can be understood from the energy balance: in the normal (but not in the superconducting) state, an external field B is Pauli-screened, with an energy gain of $\chi_{\text{Pauli}}B^2/2$, where χ_{Pauli} is the Pauli susceptibility determined by the density of states, $N(E_F)$, at the Fermi level (E_F). In the Bardeen-Cooper-Schrieffer (BCS) theory the pairing energy is $N(E_F)\Delta^2/2$, where Δ is the superconducting energy gap. Hence, when B_P exceeds Δ , it is energetically favourable to destroy superconductivity.

A superconducting state that can screen an external field is not subject to Pauli-limiting pair breaking, such as in some triplet superconductors³. Recently discovered Ising superconductors (IS) are also not Pauli limited for in-plane magnetic fields. Their spin-orbit coupling (SOC) splits the bands and leads to electron spins that are perpendicular to the 2D plane with the spins of the SOC-partner states aligned antiparallel. One consequence is that the Pauli screening for in-plane magnetic fields is effected by the SOC split states that are removed from the Fermi level and therefore not affected by superconductivity⁴. As long as both the Zeeman shift and Δ are small compared to the SOC splitting (i.e., many hundreds of Tesla in NbSe₂) the Pauli pair breaking is absent^{5–12}. This is not unrelated to the resilience of triplet superconductivity to magnetic fields: in IS half of the Cooper pairs are described as singlet + triplet, and half as singlet - triplet⁴. The absence of inversion symmetry is a prerequisite for SOC to lift the Kramers' degeneracy. In the canonical IS, NbSe₂, this is satisfied in a monolayer, but not in a bilayer or bulk, preventing IS in bulk NbSe₂⁵. In this study we report experiments and theory that clearly demonstrate that bulk layered compounds containing NbSe₂ can exhibit Ising protection^{13,14}.

Results

Figure 1a, b illustrates the two compounds that we synthesize and investigate, (LaSe)_{1.14}(NbSe₂) and (LaSe)_{1.14}(NbSe₂)₂, which we refer to as 1T1H and 1T2H, respectively. Previous work has shown that both are superconducting with $T_c = 1.23$ K and 5.7 K, respectively, and they both exhibit an in-plane critical field $B_{c2//ab}$ that strongly exceeds the Pauli limit B_P^{13} . For (LaSe)_{1.14}(NbSe₂)₂ the ratio of $B_{c2//ab}/B_P$ is ~ 5, for (LaSe)_{1.14}(NbSe₂) it is ~10 – a clear signature of IS in bulk compounds. Their fundamental structural units are monolayers of NbSe₂, where Nb atoms are in a trigonal prismatic coordination with Se, and LaSe with a rock salt structure. A stoichiometric compound of NbSe₂ and LaSe forms when one of the constituent substructures distorts to accommodate the lattice parameters of the second substructure, hence, they are often referred to as misfit compounds¹⁵. One unit cell of NbSe₂ has a basal plane area of 10.25 Å² ¹⁶, while one unit cell of LaSe has an area of 9.19 Å² ¹⁷; their ratio is larger than 1 and leads to the two misfit compounds we investigate.

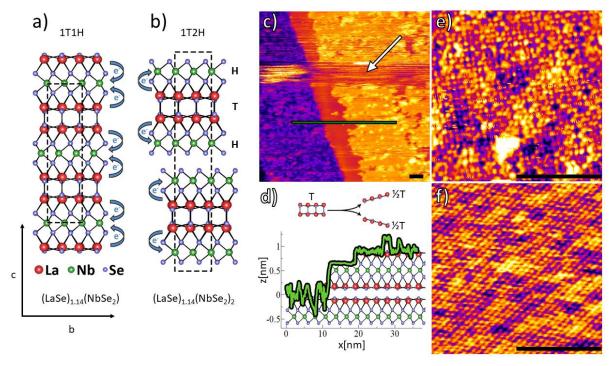


Figure 1 – Bulk misfit compounds that exhibit Ising superconductivity. Crystal structure of (a) 1T1H and (b) 1T2H. Dashed rectangles represent the unit cell in the *bc* plane. c) Topography of three atomic terraces of 1T1H, H surface in the middle and ½T surfaces adjacent to the H surface. The white arrow indicates the area, where the scanning tip disrupted the unstable ½T layer. d) (Top) Schematic illustration of how the T layer of 1T1H can be cleaved into two ½T layers. (Bottom) Height profile along the green line in (c), imposed on the illustrated atomic structure of the H and T layers. The vertical size of the

illustrated atomic structure is to scale while the horizontal size is exaggerated for clarity. The height profile reveals the step size between the NbSe₂, and ½T layer terrace is 3 Å, i.e., half the size of a full T layer. e) Topography of the cleaved ½T layer of 1T1H revealing the imperfect tetragonal lattice. The large white spot corresponds to the residue of the removed top ½T surface and dark spots correspond to vacancy sites. f) Topography of the H layer of 1T1H with hexagonal lattice after removing the unstable ½T layer by continuously sweeping the area with STM tip. The periodicity of the imaged T and H lattices is also visible in the Fast Fourier Transform of (e) and (f) along with more intricate details stemming from the misfit properties, described in Extended data Figs. 1 and 2. Scalebars in c), e) and f) represent 5 nm.

In 1T2H, the H-T-H (NbSe₂-LaSe-NbSe₂) trilayers are vertically stacked via van der Waals (vdW) interactions¹⁵, whereas 1T1H forms a fully 3D ionocovalent crystal¹⁸. This is corroborated by our first-principles calculations of the cohesive energy. The calculated energy to split the isolated T layer into two ½T layers (Fig. 1d) is $E_T \approx$ -56.8 meV/Å², which is lower than the cohesive energy $E_{T-H} \approx$ -100 meV/Å² that binds the T and H layers. This prediction is consistent with our scanning tunnelling microscopy (STM), illustrated in Fig. 1 c-f, where the ½T layer is seen to be much less stable than the H surface layer. Indeed, we were unable to image the ½T layer unless the sample was cleaved at liquid nitrogen temperature and subsequently scanned in-situ in ultra-high vacuum.

The excess stoichiometry of the T layer (1.14) with respect to the H layer suggests a charge transfer of 0.57 e^- per Nb, from the T layer into the two adjacent H layers in 1T2H, with an estimated carrier density $n_{est} = 4.6*10^{21}$ cm⁻³ based on the NbSe₂ unit cell lattice parameters¹⁵. The 1T2H Hall coefficient at 300 K is $R_H \approx + 1.45 \pm 0.1*10^{-9}$ m³/C (Fig. 2 a), corresponding to a charge carrier density $n \approx 4.7*10^{21}$ cm⁻³. The positive sign is consistent with transport occurring via the Nb 4d hole pocket and the measured carrier density agrees with our estimate above, and prior measurements¹⁴.

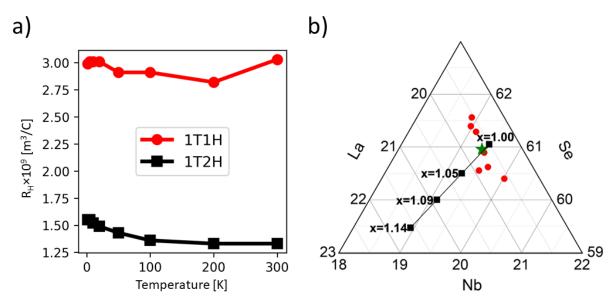


Figure 2 – Hall measurements and stoichiometry. a) Hall coefficients of 1T1H (red dots) and 1T2H (black squares). b) Projections of energy dispersive X-ray spectroscopic analyses for La_xSe_{1.14}NbSe₂ crystals in the La–Nb–Se ternary diagram (atomic percentages). Red dots are the mean analyses of each crystal while the green star represents the mean value for the seven crystals. The black line represents the theoretical compositions calculated between the vacancy-rich (x=1) and the vacancy-free (x=1.14) poles. The mean experimental composition lies on this line and evidences an amount of La vacancies around 11.1%. More details in Extended data Table 1.

For 1T1H, the same arguments suggest that each H layer accepts 1.14 e⁻ per Nb from the two adjacent ½T layers. However, based on the formal oxidation states, each Nb can accept a maximum of 1 e⁻, leaving 0.14 e⁻ per La within the T layer. First-principles calculations confirm these considerations and show the Nb 4d hole band at the K-point is occupied (Extended data Fig. 3a), while the partially filled La 5d electron band crosses the Fermi level at the Γ -point (Extended data Fig. 3b). Hall measurements indicate mobile holes with a Hall coefficient $R_H \approx + 3*10^{-9}$ m³/C (Fig. 2 a), in contrast with the negative Hall coefficient expected if transport occurs via the LaSe conduction band. The carrier density at 300 K from the Hall coefficient is n $\approx 2*10^{21}$ cm⁻³, or 0.75 e⁻ per Nb atom. Though this is larger than the carrier density in the 1T2H structure, it is well below the maximum possible doping of 1 e⁻ per Nb.

While evidence of La with a formal valence lower than 3 has been found in other layered compounds¹⁹, a more plausible explanation of this discrepancy are deviations from stoichiometry. Point defects that act as acceptors can compensate the charge transferred into NbSe₂ and lower the effective doping. We consider the role of La vacancies in LaSe, motivated by the fact that

prior work on other misfit compounds identified cation vacancies^{20–22}. Figure 2b summarizes our EDS measurements of sample composition. Despite small variations over different crystals, the mean composition indicates a La vacancy concentration of 11.1%, suggesting that 0.76 electrons were transferred per Nb atom, in agreement with the Hall data.

Our ARPES measurements of 2H-NbSe₂, 1T2H and 1T1H (Fig. 3 a-c) show the Nb 4d states crossing E_F shift towards higher binding energies, as expected with electron doping of the H layer. Furthermore, the Nb 4d hole band is not completely filled in the 1T1H structure, consistent with our positive Hall coefficient.

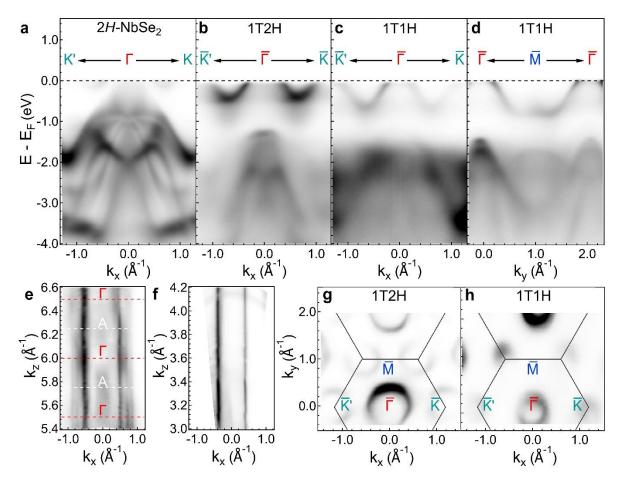


Figure 3 – ARPES measurements of bulk NbSe₂ and misfit compounds. (a) ARPES map along the $K'-\Gamma-K$ high-symmetry line of the hexagonal Brillouin zone of 2H-NbSe₂. $h\nu=106$ eV. (b) ARPES map along $\overline{K'}-\overline{\Gamma}-\overline{K}$ of 1T2H [(LaSe)_{1.14}(NbSe₂)₂]. (c-d) ARPES maps along $\overline{K'}-\overline{\Gamma}-\overline{K}$ and $\overline{\Gamma}-\overline{M}-\overline{\Gamma}$ of 1T1H [(LaSe)_{1.14}(NbSe₂)]. $h\nu=150$ eV for both misfits. (e-f) Out-of-plane momentum (k_z) dispersions at $k_y=0$ Å-1 and E_F for NbSe₂ and 1T2H. $h\nu=100$ -160 eV and $h\nu=20$ -60 eV for 2H-NbSe₂ and 1T2H, respectively. An inner potential ν_0 of 17.6 eV²³ and a work function value of 5.9 eV²⁴ were used for

converting the photon energies into k_z values within a free-electron final state approximation. The integer and half-integer values of k_z correspond to the Γ points in 2*H*-NbSe₂ (out-of-plane lattice parameter c = 12.5 Å)²⁵. (g-h) Fermi surfaces of 1T2H (g) and 1T1H (h) with overlaid hexagonal surface Brillouin zones of 2*H*-NbSe₂ and associated $\overline{\Gamma}$, \overline{K} , $\overline{K'}$, and \overline{M} high-symmetry points.

At the $\overline{K}(\overline{K'})$ points, E_F lies very close to the top a Nb 4d band. In bulk 2H-NbSe $_2$ the Fermi surface around the \overline{K} points is comprised of two spin-degenerate trigonally-warped barrels formed by the Nb 4d bands 26 . These split bands are visible around $k_x = \pm 0.8$ Å $^{-1}$ in the out-of-plane momentum (k_z) map. Fig. 3 e shows their weak, but finite, k_z -dispersion at the outer Γ pocket around $k_x = \pm 0.5$ Å $^{-1}$. While this k_z distortion is substantially smaller than the one found in the bulk calculations (see Extended data Fig. 3), one should keep in mind that the penetration depth in ARPES measurements is smaller than the interlayer distance and for this reason a perfect agreement is not expected. The inner Γ pocket derived from the Se $4p_z$ orbital forms the highly three-dimensional "pancake" FS. The 2D nature of the electronic structure of the 1T2H misfit compound is evident in Fig. 3 f. Only the outer Γ pocket remains, around a reduced k_x value of ± 0.4 Å $^{-1}$, due to the electron doping, and no k_z dispersion is visible.

Using the Fermi wave-vectors, k_F^{1T2H} and k_F^{1T1H} , of the misfit's $\overline{\varGamma}$ pockets in Figure 3 g, h we estimate the charge transfer in 1T1H by comparing the areas of the FS. We obtain a ratio of the hole densities $\frac{n_{2D}^{1T1H}}{n_{2D}^{1T2H}} = \left(\frac{k_F^{1T1H}}{k_F^{1T2H}}\right)^2 \approx 0.43$. Assuming that LaSe transfers 0.57 e^- per Nb atom in 1T2H, the charge transfer obtained from ARPES in 1T1H is 0.82 e^- per Nb atom, which compares well with the 0.75 e^- obtained from the Hall and EDS measurements.

Figure 4 illustrates the calculated band structure of the monolayer NbSe₂, stoichiometric 1T2H and off-stoichiometric 1T1H with a La vacancy concentration of 12.5% comparable to the experimentally estimated value of 11.1%. For the misfit compounds, a large amount of electron doping into the H layer shifts the E_F towards the top of the Nb 4d hole band at the K point, consistent with our ARPES and Hall measurements.

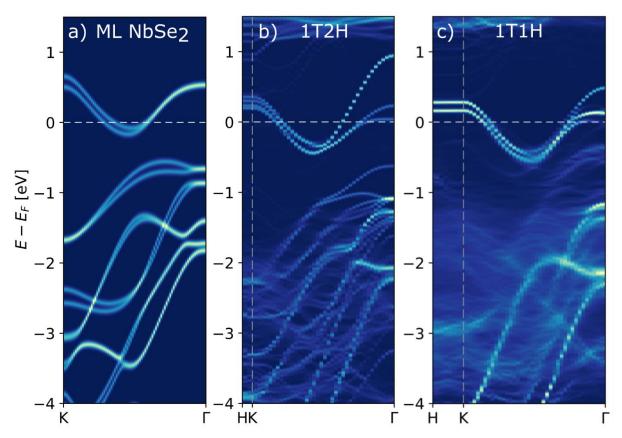


Figure 4 – First-principles calculations. Electronic structures unfolded onto the hexagonal Brillouin zone along high symmetry lines. a) Band structure for supercell calculations of bare monolayer of NbSe₂ b) 1T2H and c) 1T1H with 12.5% La vacancies.

Discussion

To provide insight into how Ising protection arises in these bulk compounds we start with the 1T1H structure. Consider a NbSe₂ bilayer, where the combination of inversion symmetry and SOC leads to a pair of doubly degenerate states at K and K', and an interlayer hopping, which we approximate as a constant, t. The magnitude of t in the 1T1H compound is different from that in the pristine NbSe₂ bilayer, since the separation between the H layers increases by ~ 5.6 Å and the interlayer tunnelling proceeds thorough the T layer rather than vacuum. We can use a 4x4 Hamiltonian to represent the spin up and down states in each NbSe₂ layer and calculate the electronic structure. By accounting for SOC, effective interlayer hopping and a Zeeman term due to an applied magnetic field one can show the Pauli susceptibility has two principal contributions: one from electrons at E_F , and the other from the states split-off by SOC that are farther away from E_F . The former term disappears in the superconducting state, the latter does not. The thermodynamic balance that determines the critical field is defined as $(\chi_n - \chi_s)B_t^2 = N(E_F)\Delta^2$, where $\chi_n =$

 χ_{Paulli} is the susceptibility in the normal state, χ_{S} is the susceptibility in the superconducting state and B_{t} is the thermodynamic critical field. For a bilayer with SOC and an effective interlayer hopping t, we get $\chi_{S}=\chi_{Paulli}\frac{\zeta^{2}}{\zeta^{2}+t^{2}}$ and by averaging over k_{Z} , for the bulk, we get $\chi_{S}=\chi_{Paulli}\frac{\zeta}{\sqrt{\zeta^{2}+t^{2}}}$. In 1T1H the spin-split bands at the K point are separated by ~150 meV (Fig. 4 c), therefore $\zeta\sim75$ meV. Now, t is $\frac{1}{2}$ of the bandwidth along the out-of-plane direction. The out-of-plane bandwidth is k-dependent, the dispersion is flat along K-H, i.e., $t\sim0$, while it is finite but small along $\Gamma-A$. We estimate the effective hopping relevant for the Ising protection as $t\sim25$ meV. This leads to an enhancement of the critical field over B_{P} by a factor $B_{P}\frac{\sqrt{\zeta^{2}+t^{2}}}{t}\sim3B_{P}$. Therefore, the 1T1H bulk crystal can be considered as an "infinite" stack of rigidly doped NbSe₂ monolayers with extremely weak interlayer hopping via LaSe layers. This Ising protection due to reduced interlayer hopping is also consistent with the conjecture by Xi et al. 5 regarding the Pauli-violation for bilayer and trilayer NbSe₂.

In 1T2H, the structure can be viewed as a stack of centrosymmetric 2H-NbSe₂ bilayers separated by LaSe misfit layers (Fig. 1 b). Here we replace t with a much larger number t_0 to account for the stronger interlayer coupling within the bilayer structure. If one assumes that $t_0 \sim 2\zeta$, as it is in bulk NbSe₂, we find the critical field is enhanced by a factor of $1.12B_P$, lower than the factor of 5 observed in our experiments. To resolve this discrepancy, we account for the fact that the LaSe layers below and above the NbSe₂ bilayer are oriented differently with respect to NbSe₂, thus breaking the total inversion symmetry in the misfit¹⁵. We emulate this effect in our theory by applying a constant energy shift of $\pm T_0$ to the upper and lower layers of the bilayer, which lifts the Kramers' degeneracy. In Fig. 4b, two pairs of NbSe₂ bands are present at the K point. Each pair is spin-split by ~ 110 meV and they are mutually shifted by ~ 50 meV at the K point (Fig. 4 b).

This results in two effects: one, which is often considered in the discussion of single monolayers²⁸, leads to a canting of the spin directions away from the c-axis. The resulting spin can be projected onto the out-of-plane direction, yielding an Ising component, and onto the *ab*-plane, yielding planar components (Rashba and Dresselhaus). The latter have no Ising protection for in-plane magnetic fields²⁸, so it is often believed that breaking the intralayer mirror symmetry is detrimental for IS.

What has been missed previously is the second effect, which helps restore the Ising protection. If we solve our 4x4 Hamiltonian including the effect of T_0 , one can show that the Ising protection is now determined by the larger of the two expressions; either $B_t = B_P \frac{\sqrt{\zeta^2 + t_0^2}}{t_0}$ or $\mu_B B_t =$

 $\sqrt{t_0^2 + (T_0 + \zeta^2)^2} - \sqrt{t_0^2 + (T_0 - \zeta^2)^2}$ (D.W., D. Agterberg, I.I.M, in preparation). In our case, T_0 ~ 25 meV at the K valley. However, since it is k dependent, we consider T_0 ~ 10 meV as a lower bound. This, combined with $\zeta \sim 55$ meV and $t_0 = 2\zeta$, yields $B_t \sim 150$ T, demonstrating that a finite energy shift due to asymmetric encapsulation provides strong Ising protection. This Ising protection is likely to be further reduced due to an admixture of planar SOC²⁸ as we discuss above, but it is unlikely to overcome the large value of B_t .

Hence, this hierarchy of energy scales of interlayer coupling, namely, breaking of inversion symmetry by the LaSe layer, SOC and the magnitude of the superconducting gap in the 1T1H and 1T2H compounds provides a consistent picture for the strong Ising protection observed in these bulk compounds. This is different from the IS in the monolayer where only the last two energy scales are present. The presence of charge doping in these misfit compounds, larger than what can be achieved with traditional substitutional doping, introduces the possibility of shifting the Fermi level to be in-between the two spin-split bands, which is predicted to give rise to topological superconductivity^{4,29,30}. It is plausible that these insights can be extended to other misfit compounds and bulk structures comprised of transition metal dichalcogenide layers where large in-plane critical magnetic fields that exceed the Pauli limit have been reported on but not identified as evidence of Ising superconductivity. This includes works on (SnSe)_{1.16}(NbSe₂)₂³¹, Ba₆Nb₁₁S₂₈³², cation intercalated NbSe₂³³ and organometallic intercalated compounds of 2H-TaS₂ ^{34,35}. Future experiments and theory work to determine the electronic and structural properties for this broad range of potential bulk Ising superconductors will be of great interest.

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Methods

Sample preparation

Single crystals of (LaSe)_{1.14}(NbSe₂) and (LaSe)_{1.14}(NbSe₂)₂ were grown by chemical vapor transport using iodine and the product of the solid-state reaction of the elemental precursors (i.e., La, Nb, Se). More details of the 1T2H sample preparation are described in ref.¹⁴. 1T1H was prepared identically except for the molar ratio which was changed to La/Nb/Se = 1.14/1/3.14.

Energy-dispersive X-ray spectroscopy (EDS)

The chemical composition of 1T1H samples was inspected by energy-dispersive X-ray spectroscopy on a JEOL 5800LV scanning electron microscope equipped with a PGT microanalyzer and operating at 20 keV. Large crystals were pasted and mechanically pressed into carbon tape to reach a good horizontal alignment and improve accuracy.

Scanning tunnelling microscopy (STM)

Scanning tunnelling microscopy data was acquired in ultra-high vacuum at a base pressure lower than 10⁻¹⁰ mbar and a base temperature of 1.14 K using a mechanically cut gold tip operated in the constant-current mode. Bias voltage of 1 V was applied to the sample, setpoint was 50 pA. The sample was mechanically cleaved in-situ at 77 K.

Hall measurements

The magnetotransport measurements were realized in a Quantum Design Physical Properties Measurement System using AC transport and resistivity options in the temperature range from 1.8 K to 300 K. Magnetic fields up to 9 T were applied in various orientations.

Angle-resolved photoemission spectroscopy (ARPES)

ARPES experiments were carried out at the CASSIOPEE beamline of the SOLEIL synchrotron radiation center, in Saint-Aubin, France. The electron detector was a Scienta R4000 hemispherical analyser with an angular acceptance of \pm 15°. The experiments were performed at T = 14 K with a linear horizontal (LH) polarization of the incident light. The total energy resolutions were between 30 meV and 80 meV depending on the measurements.

First-principles calculations

The electronic structure was calculated by means of density functional theory as implemented in the Vienna ab-initio simulation package VASP.6.2³⁶. The PAW pseudopotentials³⁶ were used with PBE exchange-correlation functional³⁷. The lattice parameters and atomic positions within the structural models were relaxed with the force threshold below 2×10⁻² eV Å⁻¹. Dispersive van der Waals energy correction³⁸ was used for the 1T2H systems. The band structure unfolding was performed using the VASPKIT³⁹. The cleavage energy calculations were obtained from total energy calculations for a series of finite-size slabs with particular surface terminations.

Acknowledgments

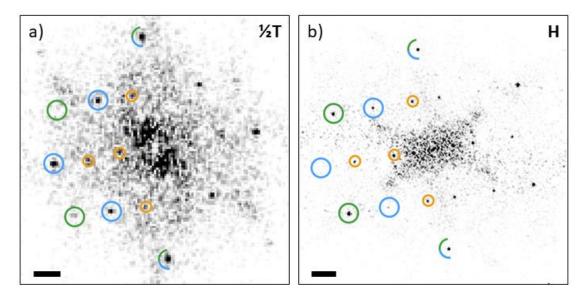
We thank V. Tkáč and J. Cordiez for assistance with the Hall measurements and chemical analysis, respectively. This work was supported by the projects APVV-20-0425, VEGA 2/0058/20, EMP-H2020 Project No. 824109, COST action CA21144 (SUPERQUMAP), EU ERDF (European regional development fund) Grant No. VA SR ITMS2014+ 313011W856. D.W. was supported by the Office of Naval Research (ONR) through the Naval Research Laboratory's Basic Research Program. I.I.M. was supported by ONR through grant N00014-20-1-2345. C.M. and G.K. acknowledges financial support from the Swiss National Science Foundation (SNSF) Grant No. P00P2_170597. M.G. and J.H. acknowledge support by the projects APVV-SK-CZ-RD-21-0114, VEGA 1/0105/20 and Slovak Academy of Sciences project IMPULZ IM-2021-42 and project FLAG ERA JTC 2021 2DSOTECH. S.S, L. C. and T. C. acknowledge financial support from the Agence Nationale de la Recherche (ANR) Project No. ANR-21-CE30-0054-02.

Author contributions

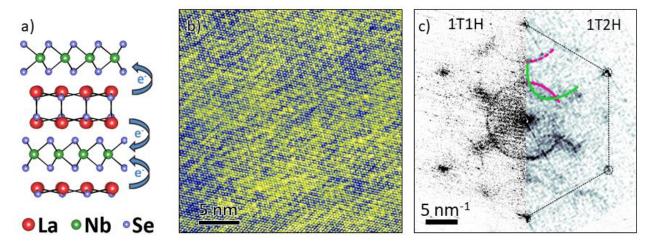
T.S., P.S. conceived the study, D.V., S.S., L.C. synthesized the samples, carried out and analysed EDS experiments, O.Š. performed and analysed the STM experiments, M.K. performed Hall measurements, T.S. supervised and interpreted the STM experiments and Hall measurements, T.J., G.K., T.C. performed the ARPES experiment with the help of P.LeF. and F.B., T.J., C.M., G.K. analysed and interpreted the ARPES data, M.G., J.H., M.C. performed the first-principles calculations, P.S., P.Sz., T.C. designed the scientific objectives and oversaw the experiments, D.W., I.I.M developed the theoretical model, T.S., D.W. and I.I.M. wrote the manuscript with input from all co-authors.

	Atomic composition (%)			Chemical formula		
Crystal (n)	Se	Nb	La	Se	Nb	La
Crystal 1 (5)	61,29(12)	19,11(7)	19,60(7)	3,140(-)	0,979(6)	1,004(6)
Crystal 2 (5)	60,6(2)	19,52(14)	19,92(10)	3,140(-)	1,012(10)	1,033(8)
Crystal 3 (7)	61,56(5)	18,90(6)	19,54(7)	3,140(-)	0,964(4)	0,997(4)
Crystal 4 (5)	61,39(12)	18,97(13)	19,64(4)	3,140(-)	0,970(9)	1,005(4)
Crystal 5 (7)	60,62(13)	19,64(9)	19,74(6)	3,140(-)	1,017(7)	1,022(5)
Crystal 6 (8)	60,89(13)	19,44(9)	19,67(10)	3,140(-)	1,002(7)	1,014(7)
Crystal 7 (5)	60,4(2)	20,0(2)	19,58(7)	3,140(-)	1,041(12)	1,018(7)
Average	60,96(13)	19,37(10)	19,67(7)	3,140(-)	0,998(7)	1,013(6)
11,1% of La vacancies	60,93	19,40	19,67	3,140	1	1,014

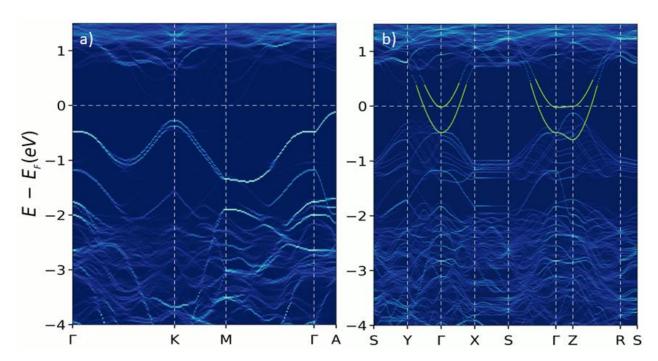
Extended data Table 1 – Chemical composition of the 1T1H samples determined by EDS. Atomic percentages with relevant errors in brackets obtained by EDS analysis for seven different crystals of 1T1H (*n* is the number of measurements on each single crystal). To calculate the chemical formula, the amount of Selenium was fixed to 3.14 according to the theoretical chemical composition (LaSe)_{1.14}(NbSe₂). The last row represents the theoretical values of the atomic percentage and chemical formula for the case of a crystal with 11.1% La vacancies.



Extended data Fig. 1 – a) and b) show the Fourier transform modulus of the topography in Fig. 1 e) and f), respectively. In both FFT images both hexagonal and square Bragg peaks are present, denoted by green and blue circles, respectively, as well as peaks corresponding to the lattice modulations due to misfit, marked by orange circles. The blue-green semicircles belong simultaneously to the square and hexagonal lattice and lie in the commensurate b direction. The fact that both square and hexagonal lattices are present in both FFT images demonstrates the surface lattice modulation induced by the underlying layer. The scalebar represents 5 nm⁻¹.



Extended data Fig. 2 – a) Crystal structure of 1T1H without the topmost $\frac{1}{2}$ T layer. b) STM topography of the H layer of 1T1H after the unstable $\frac{1}{2}$ T layer was removed by continuously sweeping the area with STM tip (30nm x 30nm, U_{bias} = 0.5 mV, I_{set} = 80 pA, T = 77 K). c) left: Fourier transform modulus of b); right: Fourier transform modulus of 1T2H topography from ref.¹⁴. When the topmost $\frac{1}{2}$ T layer of 1T1H is removed, the H surface is charge-doped only by the underlying $\frac{1}{2}$ T layer, thus mimicking the situation in 1T2H as demonstrated in a). Because the topography in b) was acquired at low bias, i.e., energy close to E_F , it is possible to resolve the quasiparticle interference patterns in its Fourier transform. The fact that this pattern is almost identical to the 1T2H case, as demonstrated in c), is well in line with the notion of the half-doped H surface layer. Though this is only a local surface effect not affecting the bulk properties of the crystal, it validates our qualitative understanding of the charge transfer between layers.



Extended data Fig. 3 – Electronic band structure along high symmetry lines for stoichiometric bulk 1T1H calculated from first principles. Band structures are unfolded to a) hexagonal and b) orthorhombic Brillouin zones corresponding to zones for NbSe₂ and LaSe crystals. The partially filled La electron bands crossing the Fermi level at the Γ -point are highlighted in green.